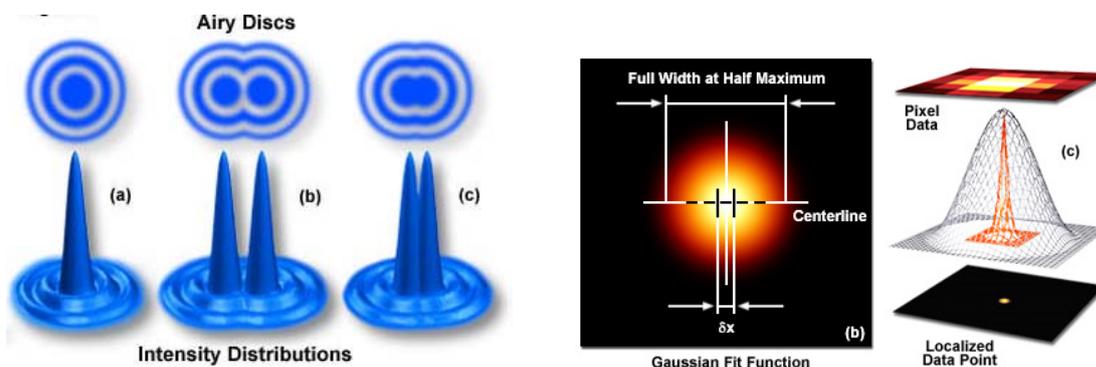


# Chapter 4

## Optical Microscopy

Instrumentation, Resolution Limits, Aberrations, Practical Resolution, Modulation Transfer Function



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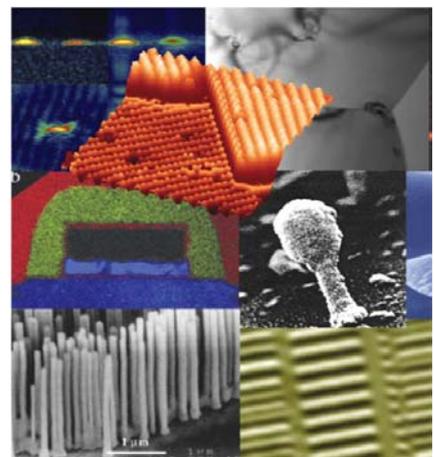
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## 4.1 Introduction

**Microscopy** = High resolution **2D** or **3D** real space *spatial imaging*, i.e., *mapping* of materials properties on a *magnified scale*.

⇒ This yields the *microscopic landscape* of properties such as morphology, mass density, chemical composition, reflectivity, absorption, luminescence efficiency, or any other property, depending on which type of response signal and probe sample-interaction is recorded as imaging signal. **Most common**: Imaging of **structure**, **morphology** or **composition** of materials.

⇒ Microscopic imaging is possible in principle for all material characterization methods described in the previous chapter. However, the **achievable spatial resolution** can be quite different due to the different used probes.




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### In this chapter:

- » The **principle approaches** and methods of microscopy are introduced and the **image formation processes** described,
- » **Resolution limits** and conditions for best resolution are derived for optical imaging microscopy and transmission electron microscopy,

- Optical microscopy (HM, PCM):**

**Visible light:** Resolution down to 200 nm ( $\approx \lambda/2$ ).

- Electron Microscopy: (TEM, SEM)**

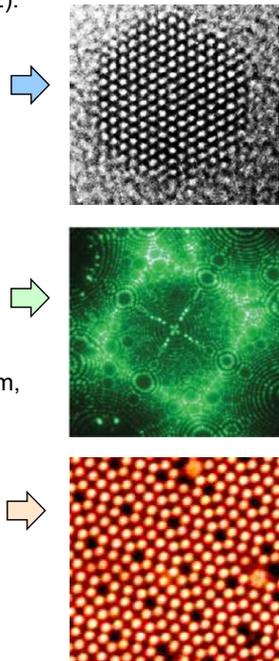
**Electron beams:** Small wavelengths  $\lambda \ll \text{\AA}$   
 TEM: Resolution of atom rows, limited by optics,  
 SEM: Resolution limited by interaction volume,

- Field ion microscopy: (FIM, FEM)**

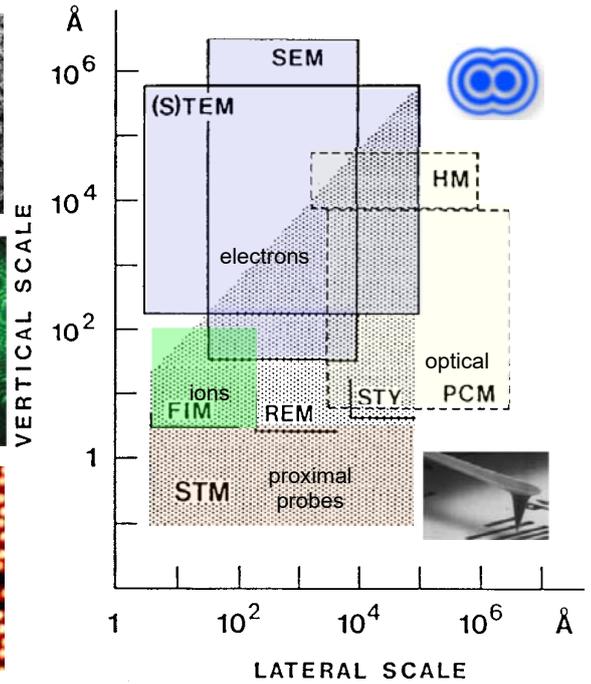
Imaging using **electrons or ions:**  
 Direct imaging of single atoms at surface steps,  
 Limited to tip-shaped samples with radius  $< 100$  nm,

- Scanning proximal probe microscopy:**

**Proximal probes** with very small tip radius and localized (near-field) interaction  
 Atomic resolution possible for STM & AFM.



## Resolution



### Type of probe determines:

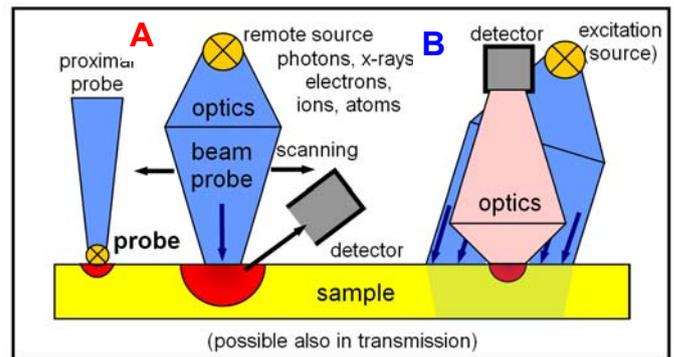
- (a) Resolution, (b) probe-sample interaction (c) information type, (d) required technical instrumentation.

## 4.2 How to Perform Imaging with High Resolution ?

### A. Scanning Microscopy = Local excitation + non-local detection

- ⇒ Excitation of a small sample volume by a tightly **focused beam probe** produced by **demagnifying optics**.
- ⇒ Global detection of the response signal from the sample by an integrating detector
- ⇒ **2D images** created by raster scanning the probe over the sample surface

**Examples:** Scanning electron microscopy (SEM), scanning optical (SOM), scanning probe (SPM), scanning x-ray microscopy (SxRM), .....



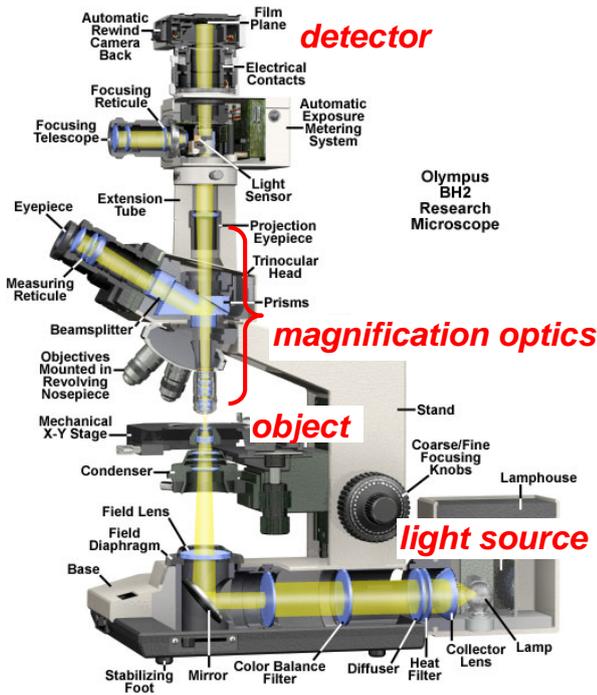
### B. Imaging Microscopy = Non-local excitation + Local detection.

- ⇒ Excitation of a large sample volume but detection only the signal from a small volume that is projected by **magnifying optics** onto the detector.
- ⇒ **2D images** created usually by **parallel detection** of many individual spots using a two dimensional pixel detector such as a photoplate, human eye, CCD camera, channel plate, etc., for which one pixel corresponds to the signal from one small sample spot.

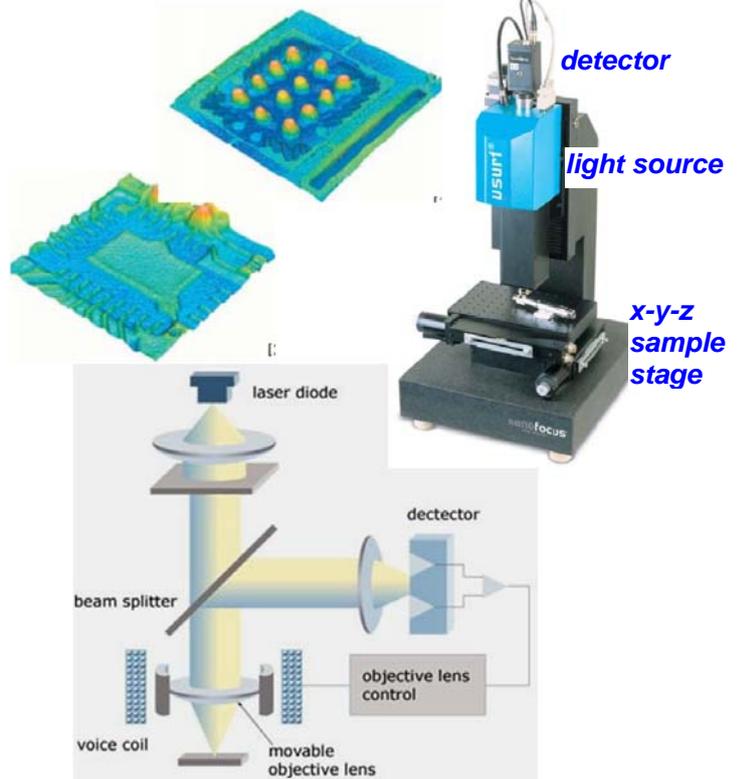
**Examples:** Transmission electron microscopy (TEM), optical imaging microscopy (OM) ...

### C. Confocal Microscopy = local excitation and local detection (Combination of both).

Example: Confocal scanning optical microscopy, ....



⇒ **Magnifying optics** required for image formation, magnification determined by focal length



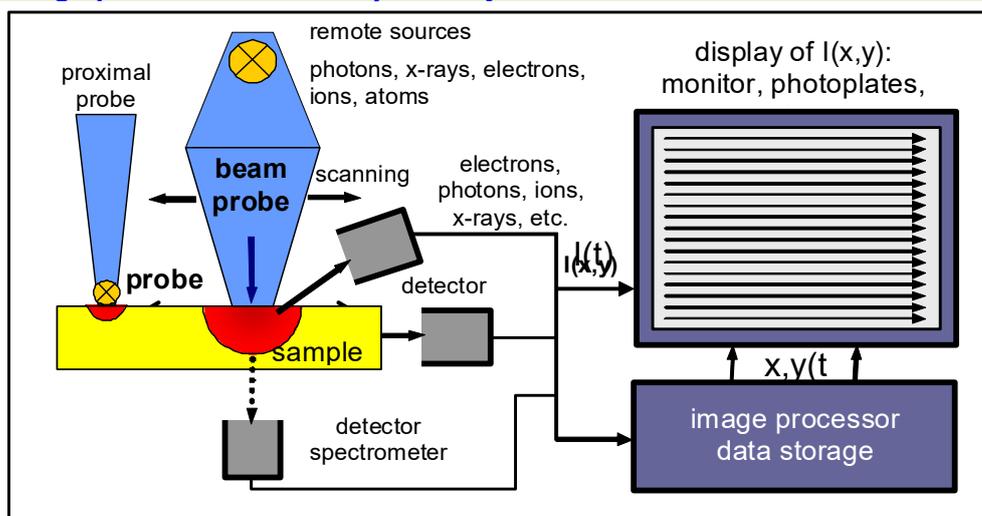
⇒ **Demagnifying optics** for probe formation, Magnification set by scan size

Note: The same type of variants exist for electron and x-ray microscopy

### 4.3 Scanning Microscopy

In **scanning microscopy**, an **image is created** by raster scanning a **small probe** that **locally** interacts/excites the sample and the **detection** of the global response signal  $I(t)$  as a function of time, where at a given time  $t$  the probe is at a different spot  $(x,y) = f(t)$  on the sample.

⇒ **Each image point** is recorded **sequentially** at a different time linked to a certain spot position !



**Magnification** = simply the ratio between display and scan size:  $M = w_{display} / w_{scan}$

The smallest **useful scan size** is limited by the probe size, i.e.,  $w_{scan} > w_{probe}$ .

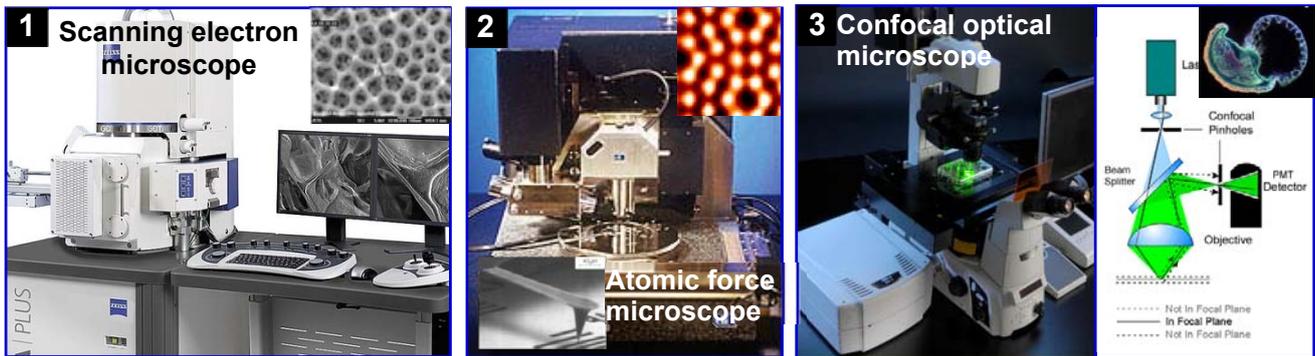
**Resolution** is determined only by the **probe size** as well as **interaction/excitation volume**.

The probe size depends on the quality of the demagnifying optics and wavelength of the probe, and the interaction volume on the probe energy and interaction strength.

# Characteristic Features of Scanning Microscopy

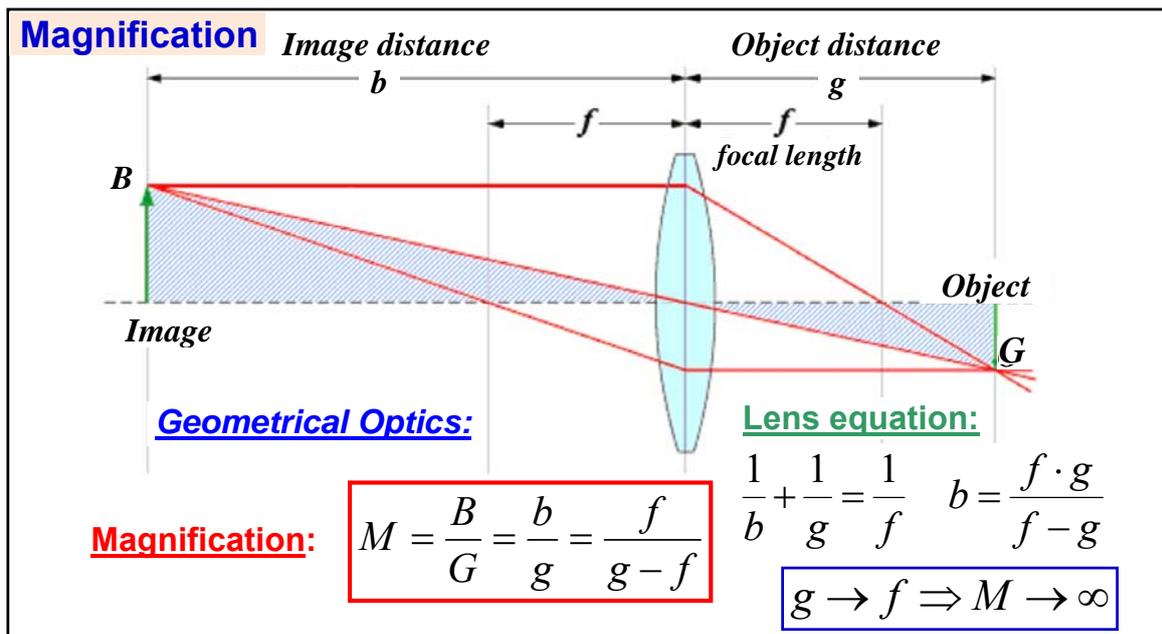
- ☺ **No magnifying optics** are required - only **simpler focusing optics** for probe formation, Thus, no demanding high resolution imaging optics are required.
- ☺ Thus, applicable to **wide range of techniques** because no imaging optics are required.
- ☺ Easy to combine with **spectroscopic techniques**. Thus, probing of many properties possible.
- ☺ **Ultimate resolution** = determined by the **probe size** and the interaction volume.  
For beam probes, the probe size is **limited by diffraction** and the **lens aberrations** of the focusing optics. The **excitation volume** depends on the probe parameters. In confocal fluorescence microscopy (STED) it can be narrowed below the diffraction limit using **non-linear optical effects**.
- ☹ **Sequential** point-by-point **recording** is slow and limits the signal integration times per pixel.
- ☺ Overall, only a **virtual sample image** is obtained that represents measured signal versus (x,y). The image must be displayed on a monitor screen, photo plate, printer, etc. .

**Variants:** Scanning Electron Microscope (SEM), Scanning Tunneling (STM) and Scanning Force Microscope (AFM), Scanning Near Field (SNOM) and Scanning Confocal Optical Microscope, ...



## 4.4 Imaging Microscopy

Local detection is achieved by creating an **enlarged image** of each sample point using **magnifying optics** of lenses or curved mirrors. The enlarged image is viewed or recorded using a 2D detector.

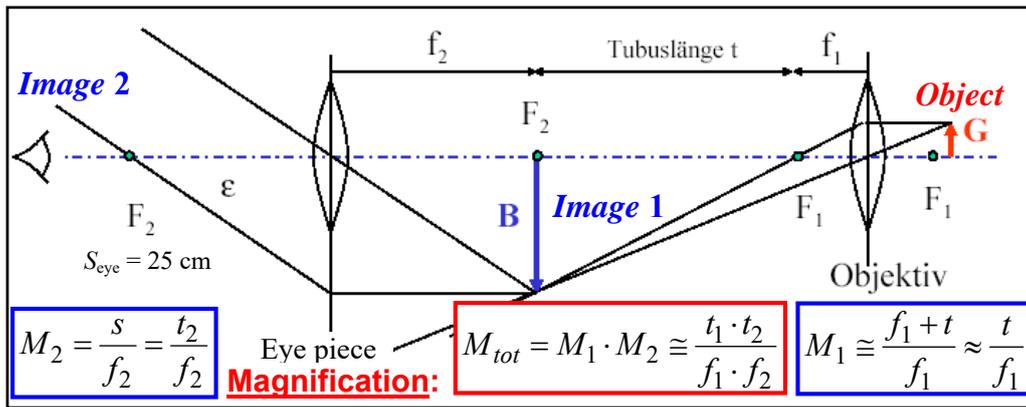


**Magnification:**  $M = B / G$  is determined by the lens equation, i.e., object and image distances  $b/g$

⇒ At any time, **each point of the image** = signal collected from **one point of the sample!**

# Magnification of a two lens optical microscope

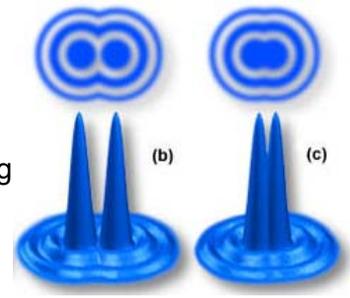
Given by lens strength  $f_i$  and tube length  $t$



⇒ The **tube length  $t$**  of optical microscopes is usually fixed (standardized to  $t = 160$  mm). Thus, different objective lenses with different  $f_i$  have to be used to change the magnification.

## Basic features of imaging microscopy:

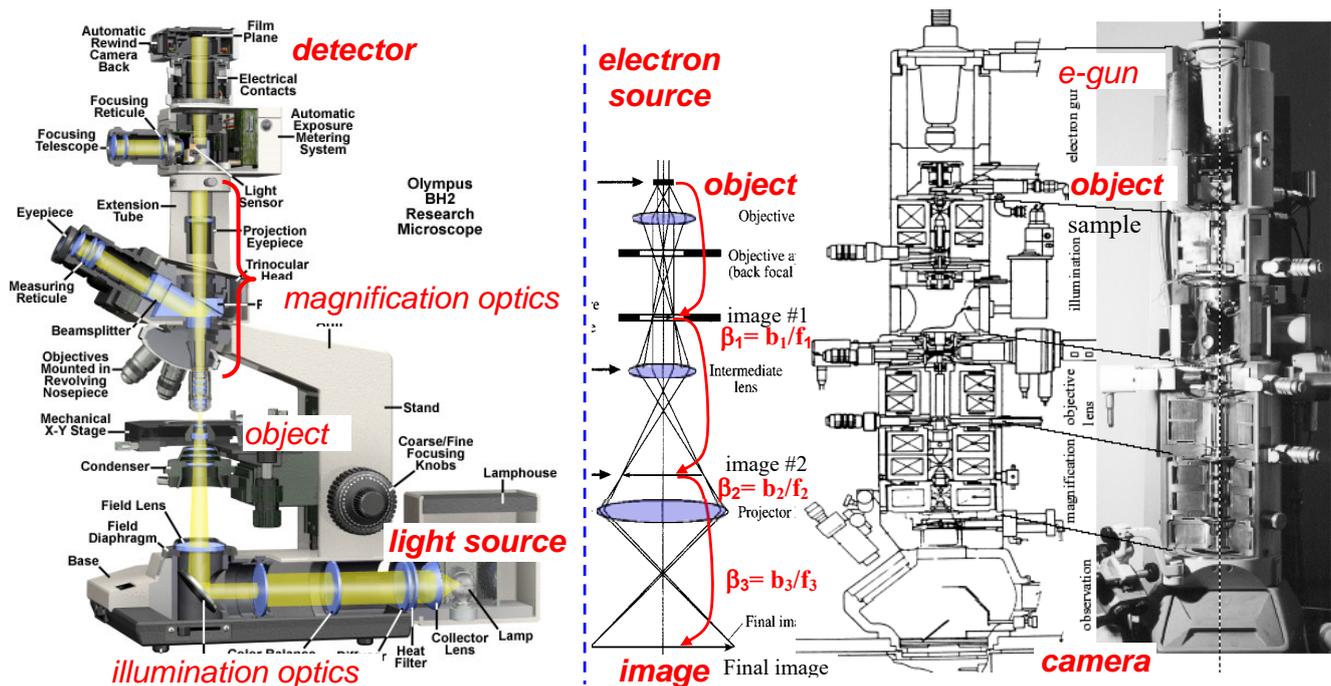
- ❖ **Resolution:** Due to *diffraction* and *lens imperfections*, the image of each point of the object is smeared out on the detector. Thus, below a certain distance  $r$ , two object points can no longer be resolved. The resolution depends on the imaging wavelength  $\lambda$  and the quality of the optical imaging system but it is **fundamentally limited** to  $r > \frac{1}{2} \lambda$ .
- ❖ **Useful magnification:** Although by using strong lenses with small  $f$  and a large tube length  $t$  the *magnification*  $M_{mag} \approx t / f$  can be made be arbitrarily large, due to the resolution limit the **“useful” magnification is finite**



# Instrumentation

For the different probes very different kinds of instrumentation, i.e., beam sources, optical elements, detectors, environmental conditions, etc. is required but the basic optical system is identical.

## Example: Optical Microscope versus Transmission Electron Microscope



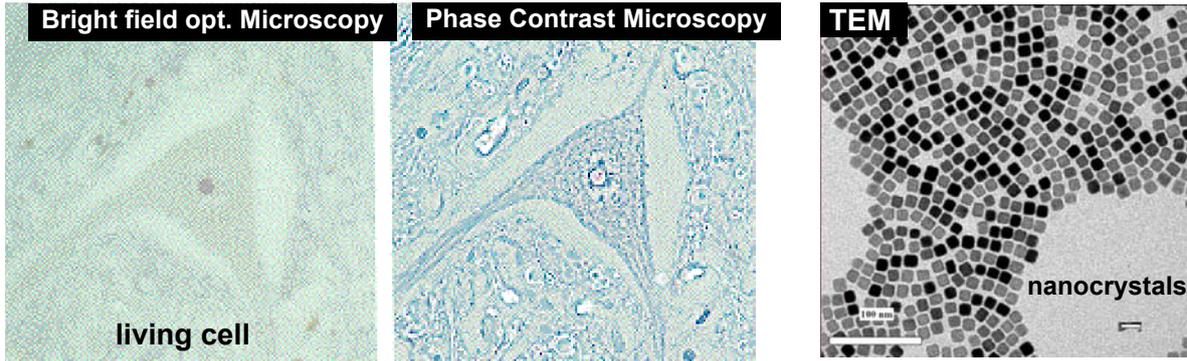
**Different elements:** (1) Beam sources: Light versus electrons, (2) Lenses: Glass versus magnetic/ electrostatic lenses, (3) Environment: Air versus vacuum column, (4) Photon versus electron detectors, (5) High voltage

# Key features of imaging microscopy

- ☺ A **real physical image** of the sample is formed within the image plane.
- ☺ Thus, all sample points are simultaneously imaged at once at the same time: This enables **real time imaging** of the whole sample at high speeds to reveal fast dynamical changes occurring within samples as a function of time.
- ☺ Long integration times per point are possible that yield high signal-to-noise ratios.
- ☹ The **resolution is fundamentally limited** by diffraction (~wavelength) .

**Variants of imaging microscopy:** Light microscopy (OM) using visible/IR/UV light, transmission electron microscopy (TEM), low energy reflection electron microscopy (LEEM), photo emission electron microscopy (PEEM), field ion microscopy (FIM), etc.

**Examples:** Left: Optical microscopy with different contrast. Right: TEM image



Different variants of imaging microscopy are distinguished according to (i) the kind of waves used for illumination and (ii) how the contrast is formed in the image.

## 4.5 Resolution of Imaging Microscopy

Generally, the lateral resolution of any microscopy system is limited by two factors:

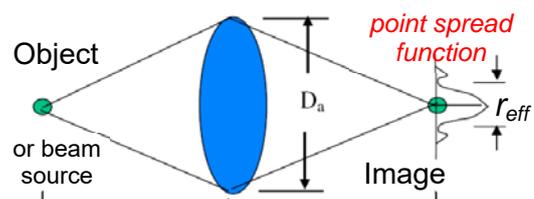
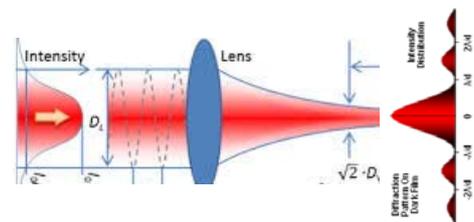
- Diffraction effects** due to the **finite entrance aperture** of the optical system = **fundamental & ultimate resolution limit** (= Abbe resolution limit, 1873)
- Image distortions/smearing** caused by **lens imperfections** (aberrations) = **“practical” resolution limit** (always > diffraction limit)



⇒ Both effects limit the resolution **independently** of the magnification of the optical system.

For **beam scanning microscopy**, both effects lead to a **broadening of the probe spot size** on the sample, with a certain minimal diameter that cannot be further reduced.

For **imaging microscopy** this leads to a **broadening of the image** of each object point, i.e., the smallest resolvable distance between to points is limited.



In sum, the total **effective resolution  $r_{eff}$**  is given by the square root summation:

$$r_{eff} = \sqrt{r_{diff}^2 + r_{aber}^2}$$

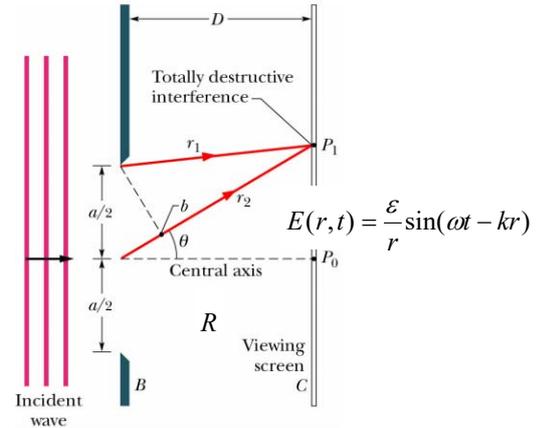
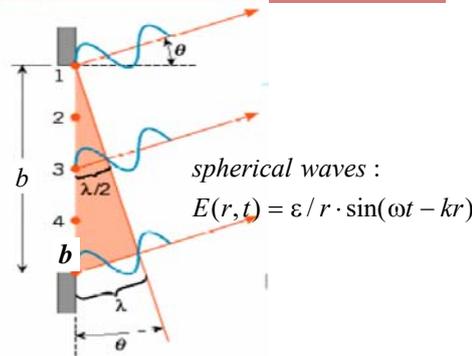
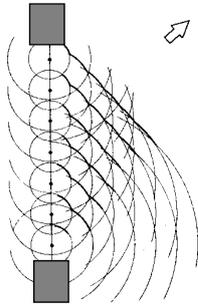
# 4.6 Abbe Resolution Limit (Diffraction)

(see, e.g., E. Hecht: "Optics")

In any optical system, the optical beam is confined by **apertures** with finite diameter and size.

At these apertures, the beams are **diffracted**. The resulting **diffraction patterns** can be derived using the **Huygens principle**, where the electric field and thus, the **intensity at a point P** away from the aperture is calculated by **summation over spherical waves** emitted with wave vector  $k=2\pi/\lambda$  from each point within the aperture of area A.

## 4.6.1 Diffraction by a Single Slit with a Width b



Emitted spherical wave :  $E(r,t) = \epsilon / r \cdot \sin(\omega t - kr)$

Total field at point P:  $E(R,\theta,t) = \int_A E(r,t) dA = \int_A (\epsilon / r) \cdot \sin(\omega t - kr) dA \cong \epsilon / R \int_{-b/2}^{+b/2} \sin(\omega t - k(R - y \sin \theta)) dy$

**Fraunhofer far-field diffraction**  
parallax approximation ( $r \gg w$ )

$r = R - y \sin \theta + (y^2 / 2b) \sin^2 \theta + \dots \cong R - y \sin \theta$  for  $y \ll R$

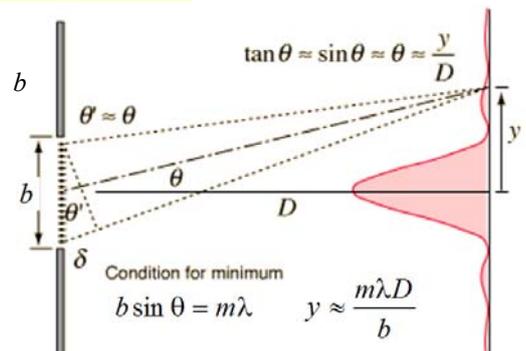
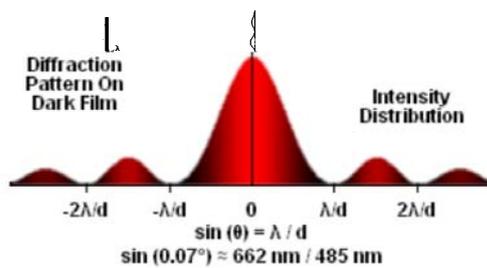
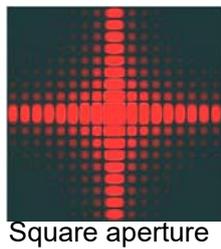
E-Field distribution:

$$\Rightarrow E(R,\theta,t) = 2 \frac{\epsilon}{R} \sin(\omega t) \frac{\sin(1/2 \cdot kb \sin \theta)}{kb \sin \theta}$$

## Far-field intensity distribution $I(\theta)$ for a single slit

$$I(\theta) = c \cdot \epsilon_0 \langle E^2(\theta,t) \rangle_t = c \cdot \epsilon_0 \langle \sin^2(\omega t) \rangle_t \cdot 2E_0^2 \sin^2(1/2 \cdot kb \sin \theta) / (kb \sin \theta)^2$$

$$I(\theta) = \langle E(t)^2 \rangle = I_0 \cdot \sin^2 \beta / \beta^2 \quad \text{with} \quad \beta = \frac{\pi \cdot b \cdot \sin \theta}{\lambda}, \quad \theta = \text{diffraction angle}$$

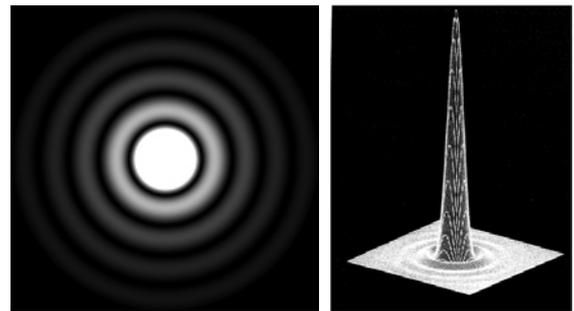
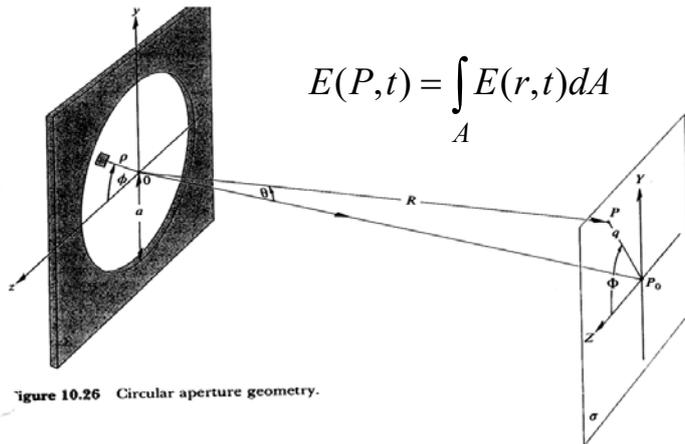


### Basic Properties:

- Central intensity maximum  $I_0$**  with series of side minima and maxima.  $I_0 = c \cdot \epsilon_0 \cdot E_0^2 = c \cdot \epsilon_0 \cdot (\epsilon / R)^2$
- Intensity minima** occur at:  $\sin \beta = 0 \rightarrow \beta = m \cdot \pi \rightarrow \sin \theta = \lambda \cdot m / b, m = 1,2,3,\dots$
- Phase difference** of out-most beams for these minima:  $\Delta\phi = m \cdot \lambda$  note:  $\lambda = \lambda_0 / n$  (wavelength in medium)
- Broadening of the image** = Separation between the central maximum and the first side minimum:  $\sin \theta_1 = \lambda / b$
- Stronger diffraction**, i.e., larger angles and broader image **the smaller the slit width  $b$** .

## 4.6.2 Diffraction by a Circular Aperture (Airy disk)

Diffraction pattern of a circular aperture (=Airy disk) with diameter  $D$  and radius  $R_0$ :  
 Derived in a similar way by *summation over all spherical waves* emitted from the disk area



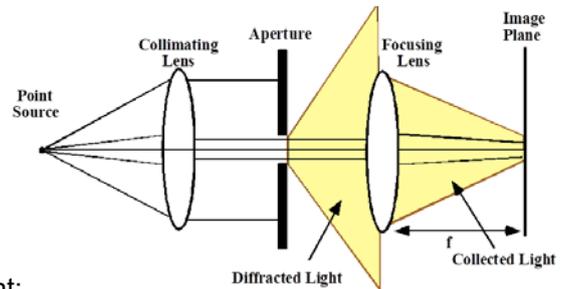
= Central intensity maximum with concentric satellite rings  
**“Airy intensity distribution”**

### The Airy intensity distribution

is described by **first order Bessel function  $J_1(x)$**  :

$$I(\theta) = I_0 \cdot 4 \cdot J_1^2(\beta) / \beta^2$$

where  $I_0$  = primary intensity,  
 $J_1(\beta)$  = 1<sup>st</sup> order Bessel function with the argument:  
 $\beta = 2\pi nR \cdot \sin\theta / \lambda$ , where  $R$  = aperture radius,  
 $\lambda$  = wavelength,  $n$  = refractive index, and  $\theta$  = diffraction angle (observation angle).



## Properties of the Airy intensity distribution:

**Intensity-Minima** are given by zeros of the Bessel function  $J_1(\beta) = 0$  occurring at

$$\beta = 3.83, 7.01, 10.17, \dots = 2\pi nR \cdot \sin\theta / \lambda \rightarrow \theta_{\min,i} = m_i \lambda / D \text{ with}$$

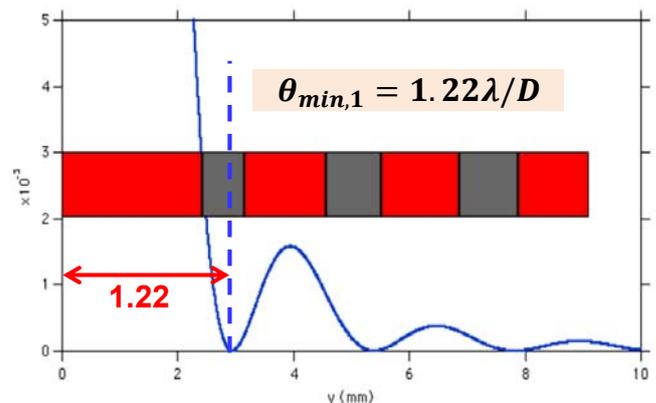
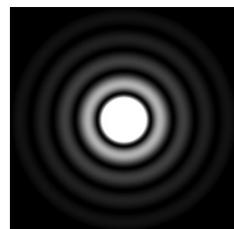
$$m_i = 1.22, 2.23, 3.24, \dots \quad (D = 2R)$$

### Radius of the 1<sup>st</sup> diffraction ring (Airy disk)

focused by a lens with a focal distance  $f$  onto a screen

$$r_{\text{Airy}} = 1.22 \lambda f / D$$

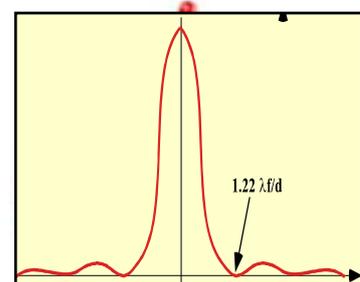
~ smallest achievable focal spot of a laser beam with diameter  $D$  and wavelength  $\lambda$



For each minimum, the path difference

between outer beams is:  $\Delta\phi_{\min,i} = m_i \lambda$  Eq. (1)

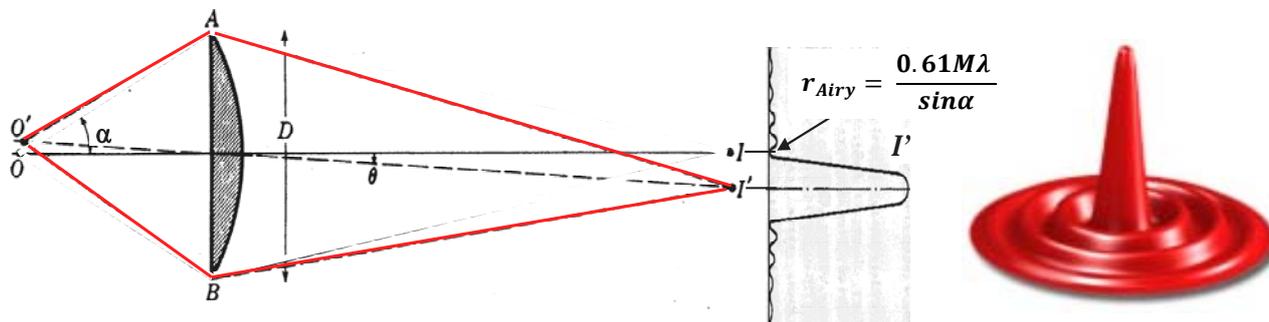
Ring	Circular aperture	Single slit
	$m$	$m$
Central maximum	0	0
First dark	1.220	1.000
Second bright	1.635	1.430
Second dark	2.233	2.000
Third bright	2.679	2.459
Third dark	3.238	3.000
Fourth bright	3.699	3.471
Fourth dark	4.241	4.000
Fifth bright	4.710	4.477
Fifth dark	5.243	5.000



**Note:** In a medium with refractive index  $n$ , the wavelength  $\lambda = \lambda_0 / n$  is shorter, accordingly:  $\theta_{\min} = \theta_{\min,vac} / n$   
 i.e., the intensity minima are more closely spaced, i.e., the diffraction broadening reduced!

## 4.6.3 Diffraction broadening in the Image: Point Spread Function

For a microscope, the resolution is generally described by the so-called **point spread function** that is the intensity distribution of the **image of an ideal point source** produced by an optical system. Its shape characterizes the performance of a microscope system and **includes all broadening factors** caused both by **diffraction** at the entrance aperture **as well as** by **lens aberrations**.



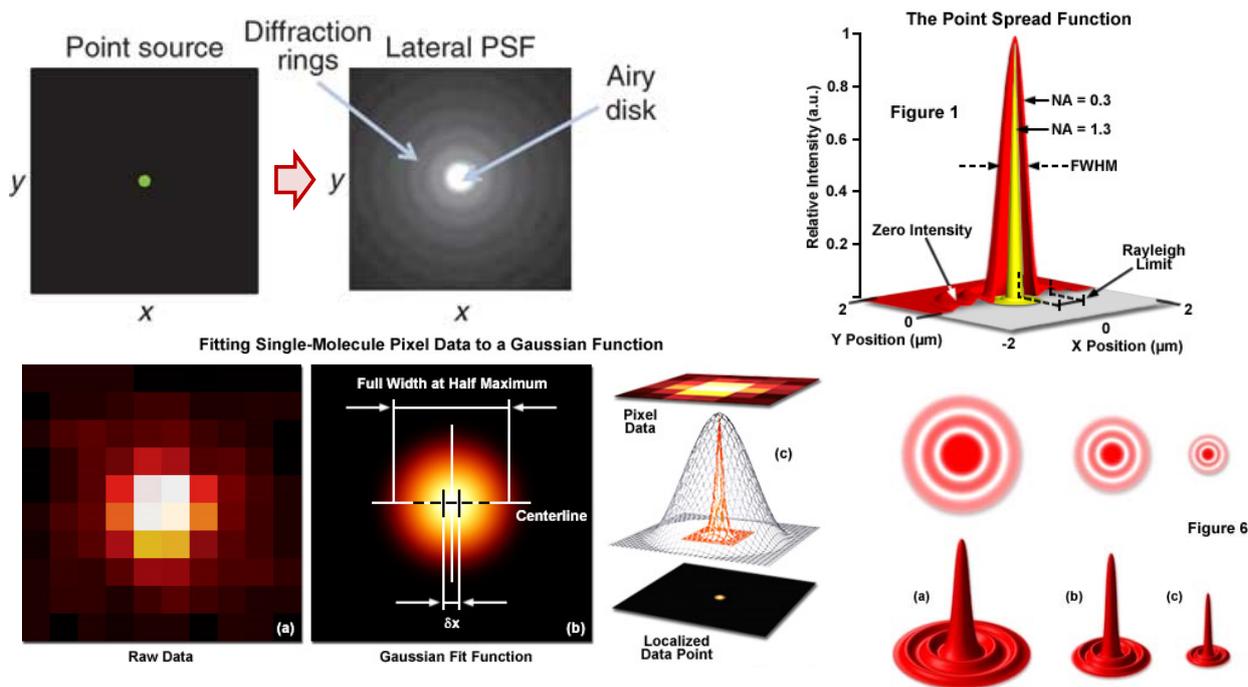
For a **perfect diffraction-limited optical system** without any lens aberrations, the lateral **point spread function**, i.e., radial intensity distribution  $I(r)$  of a point source is the **Airy diffraction pattern**, which in the paraxial approximation (small angles) is:

$$I_{Airy}(r) = I_0 \left[ \frac{2J_1(2\pi \cdot \sin\alpha \cdot \frac{r}{M\lambda})}{2\pi \cdot \sin\alpha \cdot \frac{r}{M\lambda}} \right]^2 \gg \text{Radius of the 1st Airy disk: } R_{Airy} = 0.61 M \frac{\lambda}{\sin\alpha}$$

where again  $J_1$  is the first order Bessel function,  $\alpha$  is the aperture angle,  $n$  the refractive index of the medium between the object and lens,  $M$  the magnification and  $\lambda$  the wavelength of the light.

## Experimental Determination of the point spread function

The point spread function of an optical system can be **measured** directly by recording the image of a point source, for example of a single fluorescent molecule or a small nanocrystal or quantum dot.

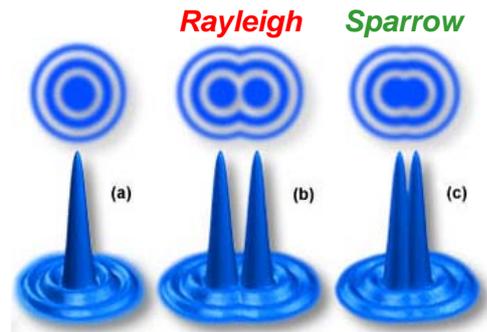


The width of the central maximum in the image, i.e., the **radius of the first dark ring** determines the ultimate microscope resolution !

## 4.6.4 Diffraction Limit of the Lateral Resolution (Abbe Limit)

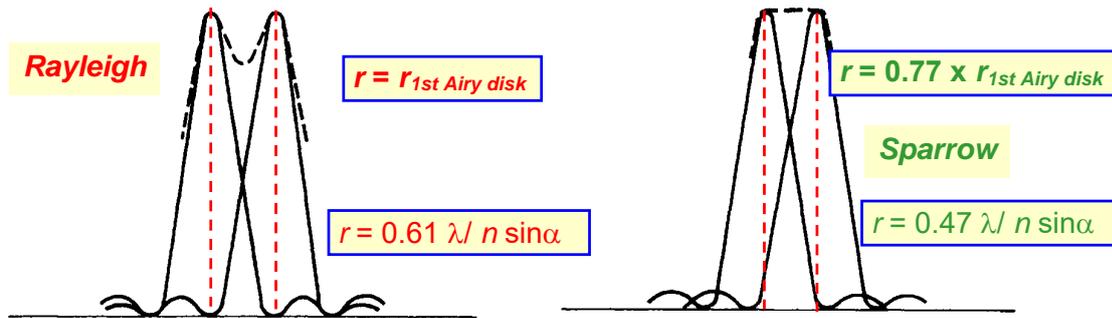
The finite diameter of the objective lens of a microscope acts as a finite size circular aperture that creates an Airy diffraction pattern in the image plane for every point of the object.

When two object points are very close to each other, the **point spread function** (Airy patterns) of their images overlap so that eventually the two points can no longer be resolved.



### (a) Rayleigh Criterion

Two point sources can be just resolved when the **central maximum** of the diffraction pattern of one point coincides with the **minimum** of the neighbouring point.



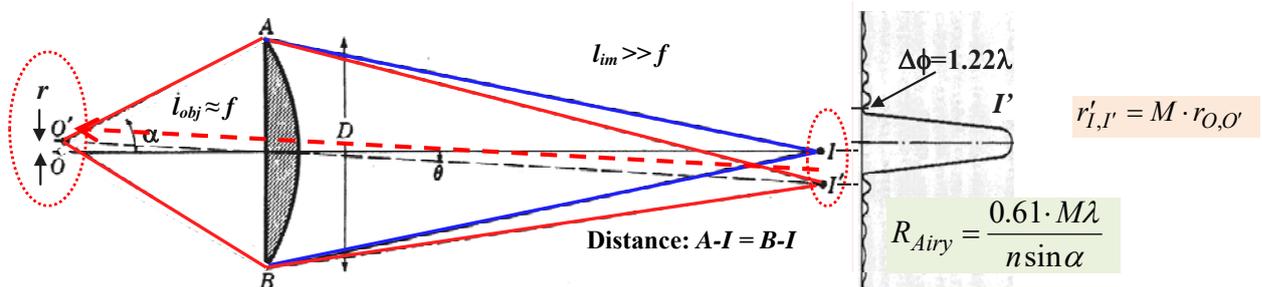
### (b) Sparrow Criterion (= Ultimate resolution limit)

Two point sources can be just resolved when there starts to exist an intensity dip between the image of the points.

## Abbe resolution limit derived by the Rayleigh criterion

In a **microscope**, the **diameter  $D$  of the objective is finite** and thus acts as a finite size circular aperture that creates an Airy diffraction pattern in the image plane for each point of the object.

As shown on the previous slides, the radius of the 1<sup>st</sup> Airy disk of the image of a point source located in the object plane at point  $O$  is  $R_{Airy} = 0.61 M \lambda / \sin \alpha$  where  $\lambda_{Airy} = \lambda_{vac} / n$  is the wavelength of light in the medium with refractive index  $n$  between the lens and the objective, and  $M$  the magnification.



⇒ Tracing back this position  $I'$  to the object plane, the image point of the Airy minimum corresponds to an object at position  $O'$  that according to the **Rayleigh criterion** can still be resolved.

As seen in the figure, point  $O'$  is separated vertically from point  $O$  by the **distance  $r = \tan \theta_{min} \cdot l_{obj}$**  where  $l_{obj} \sim f$  is its distance from the objective lens.

Since for  $\lambda \ll D$ ,  $\tan \theta_{min} \approx \theta_{min}$  it follows that  $r_{min} = 1.22 \lambda n \cdot l_{obj} / D_{len} = 0.61 \lambda / n \cdot f / R_{lens}$ .

Because  $f / R \sim 1 / \sin \alpha$

the **diffraction limited resolution** turns out to be:

$$r_{min} = 0.61 \lambda / n \sin \alpha$$

# Resolution and Numerical Aperture

The **diffraction limited resolution**  $r_{dif} = 0.61 \lambda / n \sin \alpha$

**increases** with (i) decreasing wavelength  $\lambda$ , and  
(ii) decreasing acceptance angle  $\alpha$

The denominator

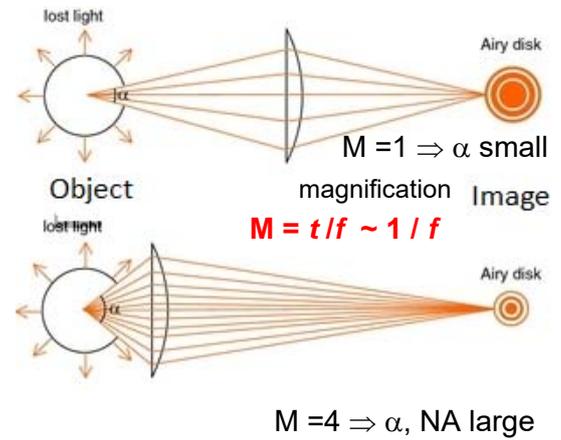
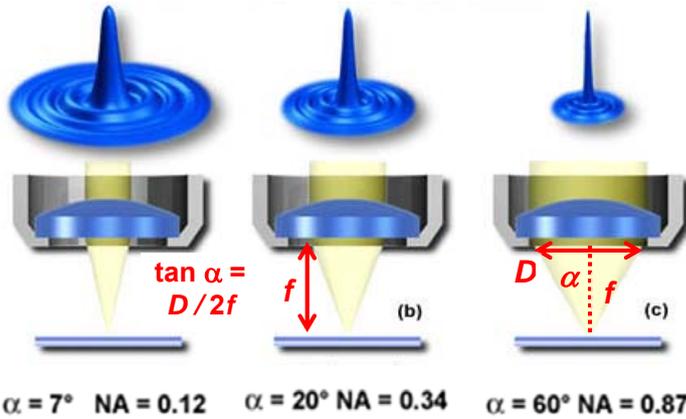
**NA = n sin α** is called **“Numerical Aperture”**

It combines the **aperture angle**  $\alpha$  of the entrance lens and the **refractive index**  $n$  of the medium between sample and lens and thus, characterizes the properties of a given optical lens system.

The **numerical aperture NA** can be increased by (a) increasing the diameter of the objective lens and/or (b) decreasing the focal length  $f$  of the lens as shown by the figures below:

⇒ **High NA** = small working distance WD » **small f-number** » **high magnification** [ $M \sim t / f$ ]

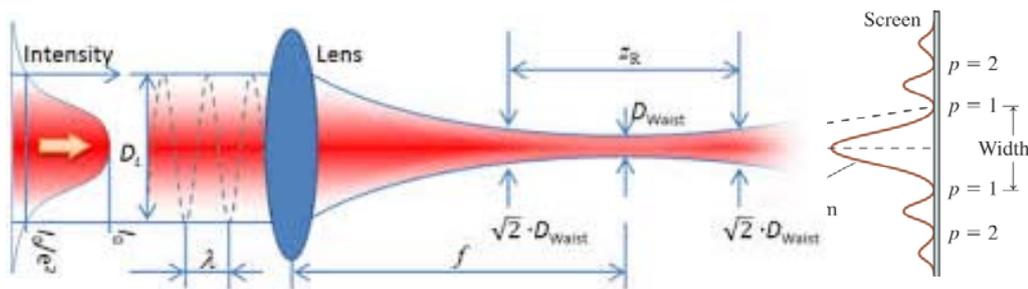
⇒ But: For  $n=1$ ,  $\sin \alpha$  is limited to  $< 1 \Rightarrow NA < 1$ .



## 4.6.5 Resolution of Scanning Microscopy: Minimal Spot Size

In scanning (laser) microscopy, imaging is performed by scanning a tightly focused laser spot over the sample surface and recording the reflected or transmitted light intensity as a function of spot position.

The resolution is given by minimal spot size, which is again limited by the diffraction at the focusing lens, or more specific, for a quasi parallel laser beam focused by the beam diameter (=beam waist).



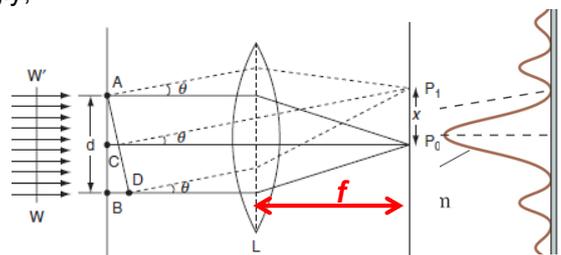
The diffraction angle of the 1<sup>st</sup> Airy disk caused by the diffraction is again given by  $\theta_{min,1} = 1.22 \cdot \lambda / Dn$ , which appears at a distance  $x$  from the optical axis with  $x = f \tan \theta$ . When the lens diameter is large,  $\theta_{min}$  is small and thus,  $x = f \tan \theta \approx f \cdot \theta \approx f \cdot NA$ . Accordingly, the diameter of the Airy disk (=spot size) is:

$$d_{spot} = 1.22 \cdot \lambda f / D$$

where the parameter  $f/D$  is called **“f” number** of the lens.

⇒ Note that for a laser with **Gaussian beam profile**, the laser does not uniformly fill the aperture, for which reason the spot size is slightly increased to

$d_{spot} = 4/\pi \cdot \lambda f / D$ . Example:  $\lambda = 680\text{nm}$ ,  $D = 5\text{mm}$ ,  $f = 5\text{mm}$  »  $d_{spot} = 1.9\ \mu\text{m}$



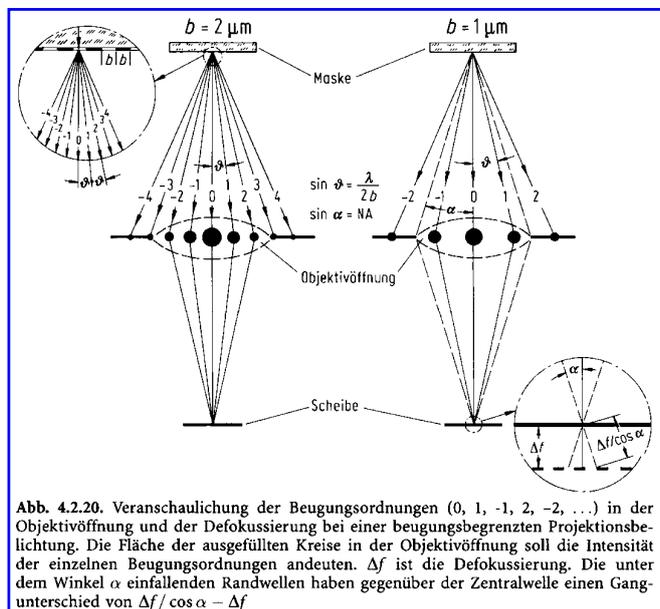
# 4.7 Modulation Transfer Function: Resolution in Fourier Space

## 4.7.1 Resolutoin for a Periodic Line Grating

For a **2b periodic line grating** with line width **b**, the angles  $\varphi_i$  of the **diffraction maxima** are given by:

$$\sin \varphi_i = i \cdot \lambda / (2 n b) \quad \text{with } i \dots \text{ diffraction order, } n \dots \text{ refractive index, } \lambda \dots \text{ wavelength}$$

If the grating is imaged with an objective lens with a capture angle  $\alpha$ , only diffracted beams with  $\varphi_i < \alpha$  can contribute to the image formation.



### Resolution Criterion:

The grating can only be resolved if at least the **1<sup>st</sup> order diffracted beams** are still collected by the lens ( $\varphi_1 \leq \alpha$ ).

**Condition:**  $\sin \varphi_1 = \lambda / (2 n b_{\min}) = \sin \alpha$

$\rightarrow R = b_{\min} = 0.5 \cdot \lambda / (n \cdot \sin \alpha)$

or  $R = 0.5 \cdot \lambda / NA$

with  $NA = (n \cdot \sin \alpha)$   
(numerical aperture of the lens)

**⇒ Thus, the objective acts as a low pass filter in diffraction space !**

⇒ **Off-axis illumination** increases the resolution because more higher order diffraction maxima go through the objective.

## 4.7.2 Fourier Decomposition of the Sample Contrast

Any real space object with intensity contrast  $C_s(x,y)$  can be thought to be composed up of a **sum of sinus and cosine functions of various wavelengths  $\lambda$**  or **wave vectors  $k = 2\pi/\lambda$** , with the **amplitude** of each Fourier component  **$A(k)$  or  $A(\lambda)$**

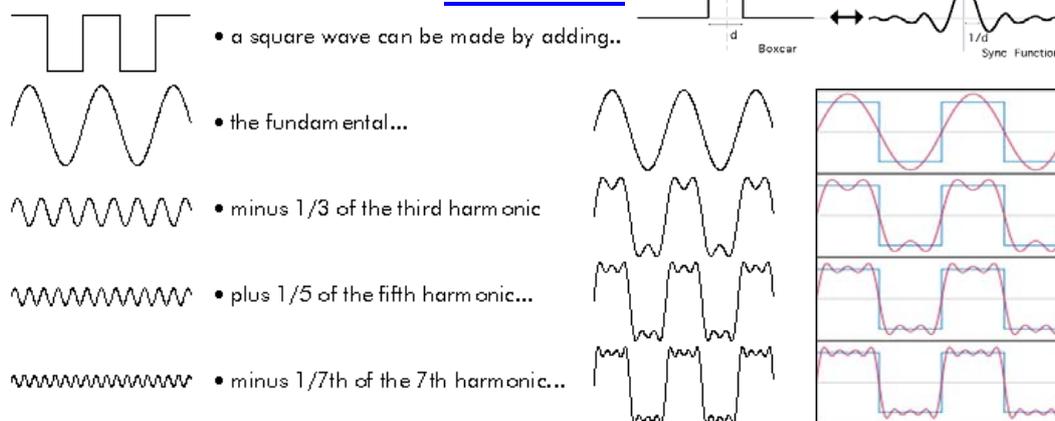
$$C(x, y) = \int \int_{-\infty-\infty}^{\infty\infty} A(k_x, k_y) e^{-i(k_x x + k_y y)} dk_x dk_y$$

(For periodic functions, the integral can be replaced by a summation)

$$C(x) = \frac{a_0}{2} + \sum_{n=1}^{\infty} A(n) \sin(k_n x) + B(n) \cos(k_n x) \quad \text{where } k_n = \frac{2\pi n}{\lambda}$$

### Example:

### Box function:



⇒ An exact 1:1 Fourier representation of  $C(x)$  requires an **infinite number of Fourier components**.

⇒ In a **real image formation process**, however, the higher order spatial frequencies are cut off, i.e.,  $A(k)=0$  for  $k > k_c$ . Thus, only a **finite Fourier series** contributes to the image and as a result, the sample contrast  $C_s(x,y)$  is no longer exactly reproduced, i.e.,  $C_{\text{image}}(x,y) \neq C_{\text{sample}}(x,y)$ .

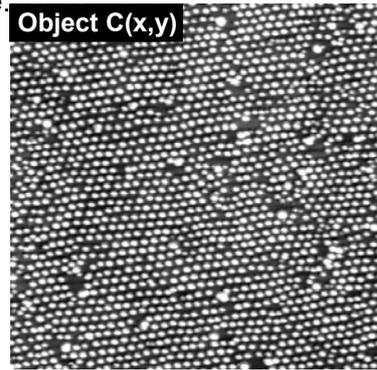
# Fourier Representation of the Sample

The Fourier space representation of the **intensity contrast**  $C_s(x,y)$  of a **real space object** is consists of the sum of plane waves with an **infinite number of spatial frequencies**  $k = 2\pi/\lambda$ . Each plane wave is weighed by a different amplitude  $A(k_x, k_y)$  (=Fourier component), i.e.

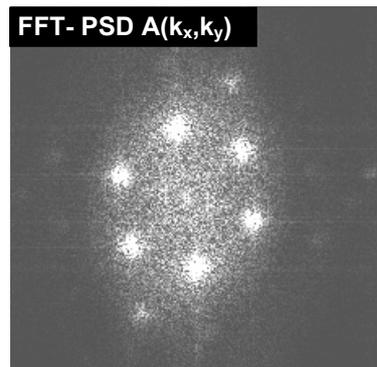
$$C(x, y) = \int_{-\infty}^{\infty} \int_{-\infty}^{\infty} A(k_x, k_y) e^{-i(k_x x + k_y y)} dk_x dk_y$$

$k_x = 2\pi/\lambda_x$  and  $k_y = 2\pi/\lambda_y$  are the **wave vectors** in x and y direction of each plane wave, which are inversely proportional to the corresponding **spatial wavelength**  $\lambda_{x,y}$ .

$A(k_x, k_y)$  is the **amplitude** of the plane wave component with wave vector  $(k_x, k_y)$ . Thus:  $C(x, y) = \text{FFT}(A(k_x, k_y))$ .



FFT  i-FFT



The **Fourier amplitudes**  $A(k_x, k_y)$  (=contribution of each frequency or wavelength to  $C(x,y)$ ) can be calculated by the **inverse Fourier transformation (iFFT)** of the object contrast function  $C(x,y)$ .

Thus:  $A(k_x, k_y) = \text{iFFT}(C(x, y))$

$$A(k_x, k_y) = \frac{1}{4\pi^2} \int_{-\infty}^{\infty} \int_{-\infty}^{\infty} C(x, y) e^{+i(k_x x + k_y y)} dx dy$$

From the knowledge of the Fourier amplitudes  $A(k_x, k_y)$  the sample structure  $C(x,y)$  can be reconstructed and vice versa.

## 4.7.3 Modulation Transfer Function

### (a) Ideal optical imaging system

For an ideal **perfect imaging process**, each spatial frequency contained in the object would be **exactly** reproduced, i.e.,  $A_{image}(k_{ximage}, k_{yimage}) = A_{sample}(k_x, k_y)$ .

Accordingly, the sample contrast

$C(x,y) = \text{FFT}(A(k_x, k_y))$  would be **exactly** reproduced in the image without any loss of information.

### (a) Real optical imaging system

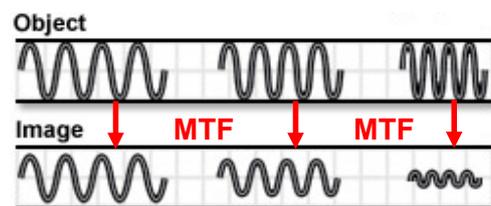
In a **real imaging system**, different spatial frequencies are reproduced differently well because **high spatial frequencies** (small period structures) are generally **reproduced less well** than **low spatial frequencies**. This means that the amplitudes  $A(k)$  of Fourier components with high spatial frequencies contained in the object are reduced by a certain factor in the image.

⇒ The ability of a lens system to reproduce different spatial frequencies of the object in the image is characterized by the **modulation transfer function MTF(f) or MTF(k)**.

### Definition of the modulation transfer function:

$$\text{MTF}(k) = \frac{\text{image contrast modulation}}{\text{object contrast modulation}} = \frac{A_{im}(k)}{A_{obj}(k)}$$

at a given spatial frequency  $k$ .

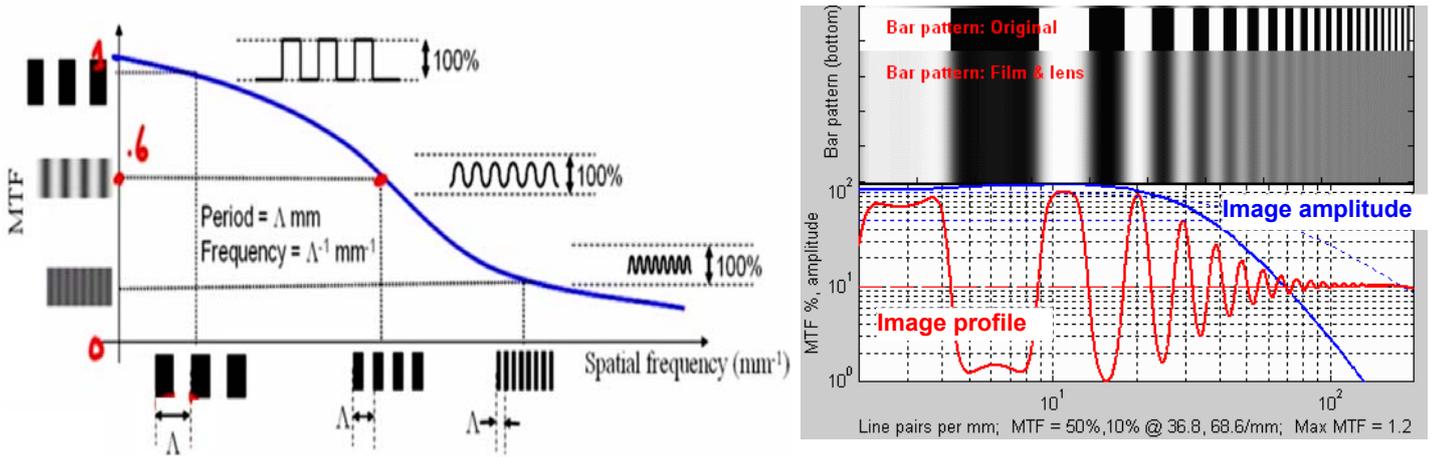


# Properties of the Modulation Transfer Function

For a perfect imaging  $MTF = 1$  at all frequencies, whereas for a real imaging system the MTF decreases with increasing wave vector, i.e., with decreasing periodicities contained in the sample. As a result, there exists a certain **maximum cut-off spatial frequency  $k_c$**  above which the MTF is zero, meaning that all high frequency information is eliminated in the image formation.

⇒ A lens system acts like a **low pass Fourier filter** such that the contributions of the higher spatial frequencies in the image are removed.

## Real modulation transfer $MTF(k)$ plotted versus spatial frequency $k = 1/\lambda$



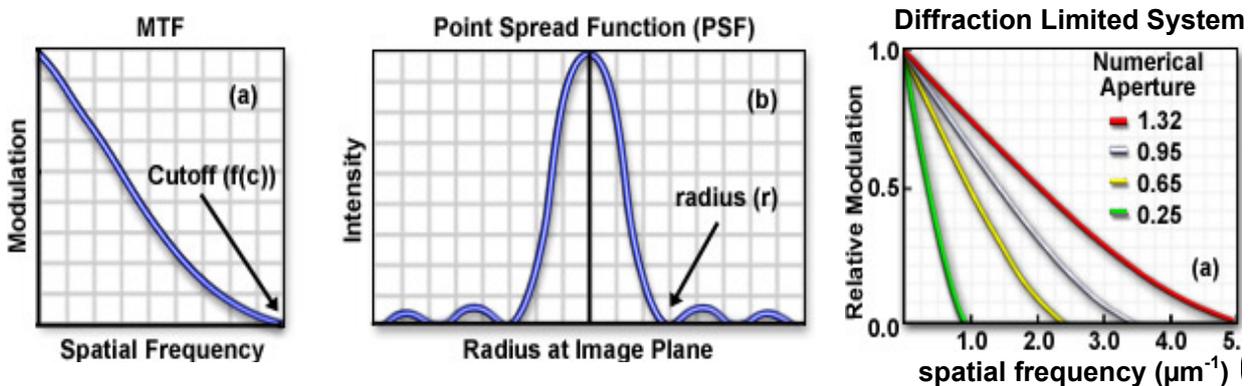
# Modulation Transfer Function $MTF(k)$ of a Diffraction Limited System

The modulation transfer function is the inverse FFT of the **point spread function** (=image of a point source).

For a **diffraction limited optical system** the MTF can be thus directly calculated from the

Airy intensity distribution as:

$$MTF(k) = 2(\phi - \cos\phi \sin\phi) / \pi \quad \text{where } \phi = \cos^{-1}(k \lambda_{illum} / 2NA)$$



- At **low spatial frequencies**,  $MTF(k) = A_{im} / A_{obj} \sim 1$ . Thus, the image contrast is equal to that of the sample. At **higher frequencies**, MTF decreases and falls to zero at the cut-off frequency  $k_c$ .
- For a **diffraction limited system**, the **cut off frequency  $k_c$**  is given by:  $k_c = f_c = 2.1 NA / \lambda$   
Thus, the **highest spatial frequency** is determined by the Sparrow limit.
- The corresponding radius of the first dark concentric ring surrounding the central peak of a point spread function is given by:  $r = 0.61 \lambda / NA$
- Objectives with low NA produce a wider point spread functions and thus, lower resolution.

## 4.7.4 Calculation of the Image using the Modulation Transfer Function

Once the modulation transfer function  $MTF(k)$  is known, the Fourier component

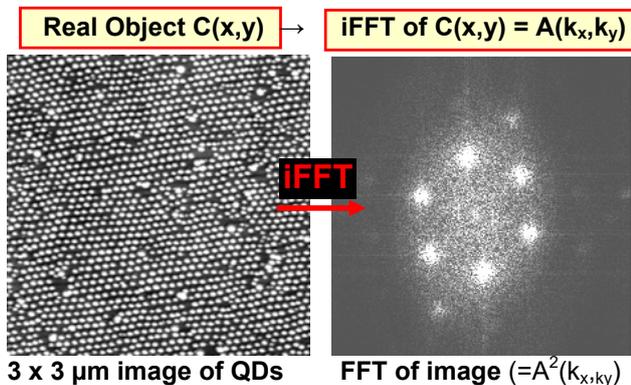
amplitudes of the image, i.e.,  $A_{image}(k_x, k_y)$  can be calculated using:

$$A_{im}(k) = MTF(k) \cdot A_{obj}(k)$$

From the resulting  $A_{image}(k_x, k_y)$ , the **real space image**  $S_{image}(x, y)$  can be calculated by back transformation of  $A_{image}(k_x, k_y)$  into real space:

$$S_{im}(x) = iFFT[A_{im}(k)] = iFFT[MTF(k) \cdot A_{obj}(k)]$$

### Example: Influence of the modulation transfer function on the image

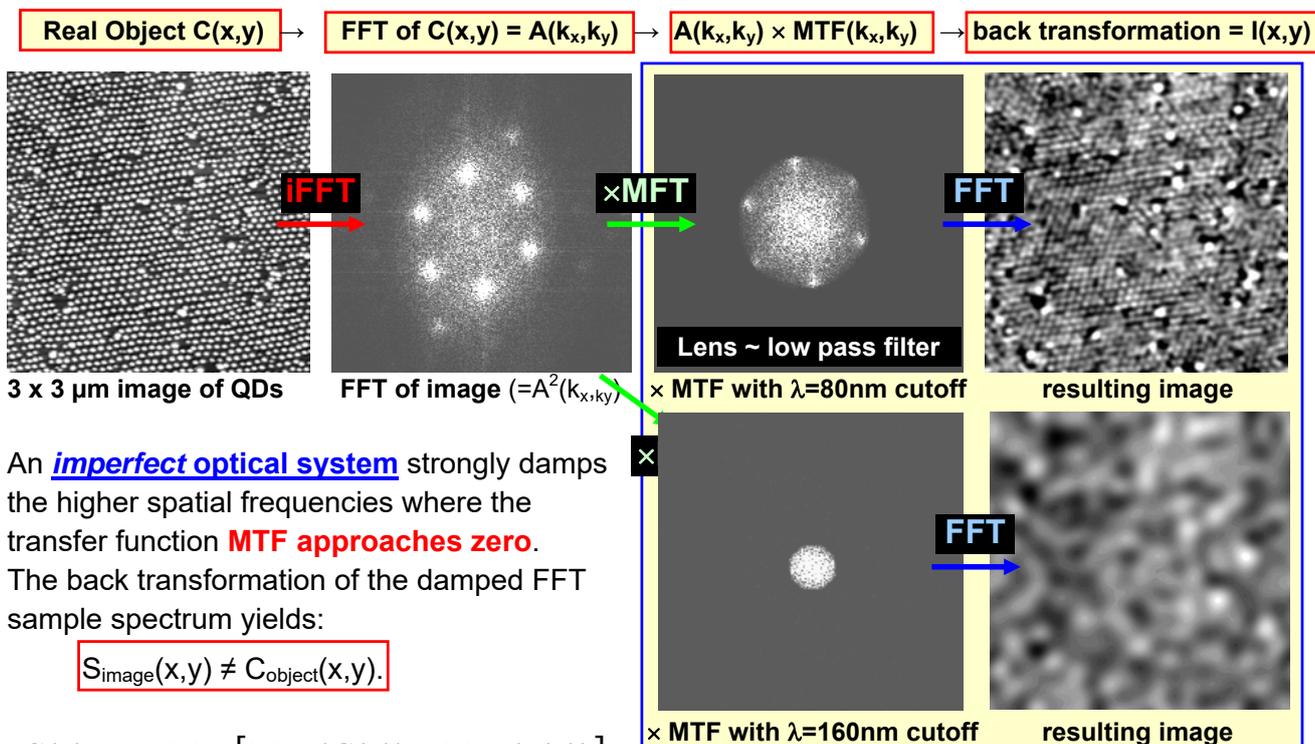


The contrast modulation of the object at a spatial frequency  $k$  is given by the corresponding **Fourier amplitude**  $A(k_x, k_y)$  of the object contrast  $C(x, y)$ .

This amplitude is obtained by the Fourier transformation of the object contrast distribution  $C(x, y)$  or  $I(x, y)$  that represents the real space structure of the object

A **perfect optical system** with infinitely high resolution would have a modulation transfer function of **MTF = 1 (unity) for all spatial frequencies**. Only then, it follows that  $I_{image}(x, y) = I_{object}(x, y)$  !!!

## Image formation calculated using the modulation transfer function



An **imperfect optical system** strongly damps the higher spatial frequencies where the transfer function **MTF approaches zero**. The back transformation of the damped FFT sample spectrum yields:

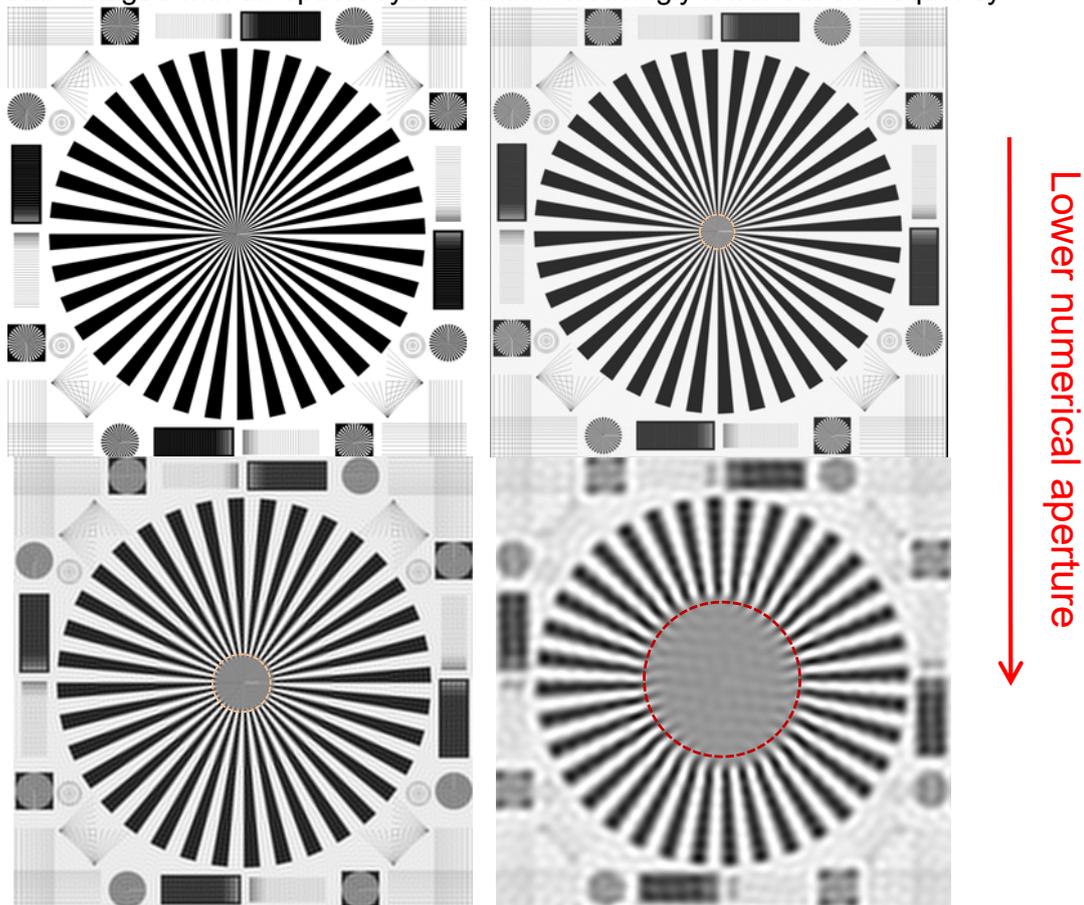
$$S_{image}(x, y) \neq C_{object}(x, y).$$

$$S(x) = invFFT[FFT(C(x)) \times FFT(r(x))]$$

The **modulation transfer function includes** both image broadening effects due to **diffraction and Lens imperfections**. A lens system **without** aberrations is termed **diffraction limited**.

## Other Example: Testpattern for Quantification of the Image Resolution

“Siemens Star” imaged with an optical system with increasingly lower cut-off frequency



### 4.7.5 Measurement of the Modulation Transfer Function

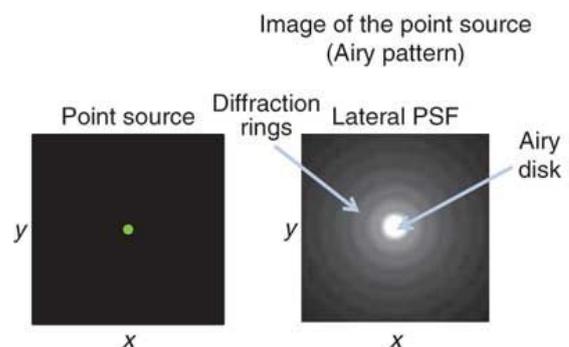
The **modulation transfer function** and **point spread function** of any optical system including a microscope can be **determined** using a **point-like light source** such as single fluorescing molecules, quantum dots or nanocrystals, or illuminated point like apertures.

#### Procedures:

- (a) The intensity distribution of the image of the point source corresponds to the **point spread function** of the microscope.

The **Fourier transform** of this intensity profile of the point source image yields the **modulation transfer function MTF(k)**

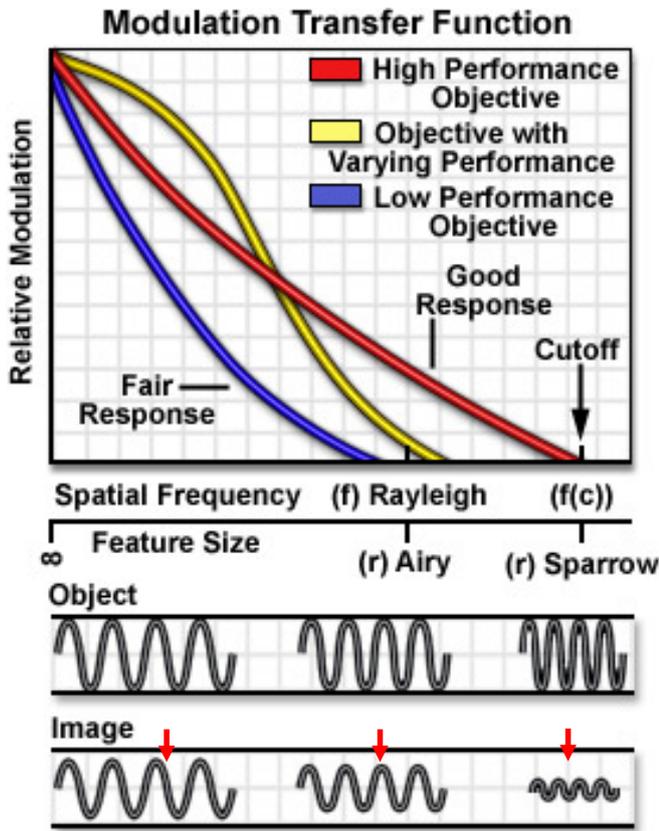
- (b) The MTF can be also **directly measured**:  
It is the diffraction pattern formed in the **back focal plane** of the objective system.



Alternatively, the intensity profile of a sharp shadow edge can be also used to determine the MTF function, again by calculating the Fourier transform of the intensity profile in the image.

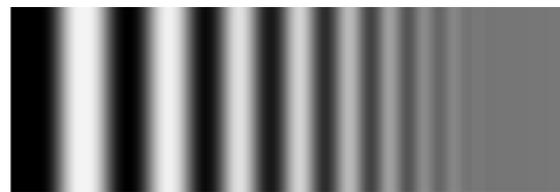
In general, for an ideal point source the **modulation transfer function** corresponds to the **diffraction pattern** formed in the **back focal plane** of the objective system, and the point spread function to the intensity distribution in the image plane.

Thus, the MTF and PSF are strictly interrelated by the Fourier transformation.



**Properties of the MTF:**

- ❖ continuously decreases with increasing spatial frequency until it reaches zero at the cut-off frequency  $k_c$  (or  $f_c$ )
- ❖ all frequencies higher than  $k_c$  are completely blocked by the optics.
- ❖ high performance objectives with large NA have a higher cut-off frequency and higher MFT values at lower frequencies.



## 4.8 Axial Resolution and Depth of Field

The point spread function is actually a **3D intensity distribution** in (x,y) as well as z (axial) direction.

**Axial Airy intensity distribution:**

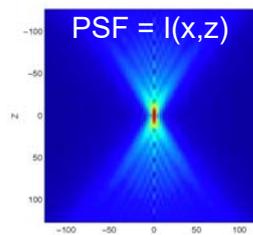
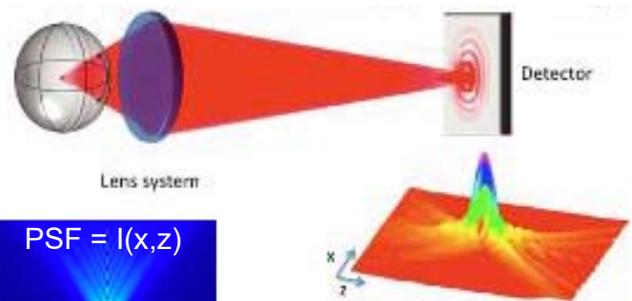
$$I_{Airy}(x=0, z) = C \cdot \frac{NA^4}{M^2} \left[ \frac{\sin(\pi NA^2 \cdot z / 2n \cdot M^2 \lambda)}{\pi NA^2 \cdot z / 2n \cdot M^2 \lambda} \right]^2$$

This means that the image of a point source is not only **broadened** in the lateral, but also **in the axial direction**, with an axial width (=1<sup>st</sup> zero of  $I(z)$ )

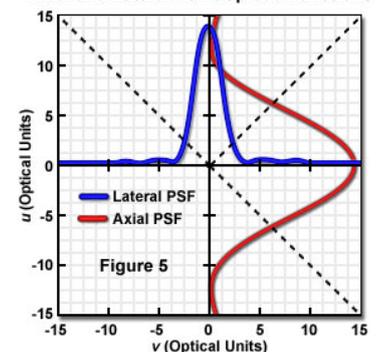
$$\Delta z_{Airy} = 2n \cdot M^2 \lambda / NA^2$$

Thus, the **axial resolution**, defined as minimal z-separation of two points that can be discriminated in the microscope when focusing through the object is limited to:

$$\Delta z_{res} = 2n \cdot \lambda / NA^2$$



Axial and Lateral Point Spread Functions



The Rayleigh Criterion for Lateral and Axial Resolution

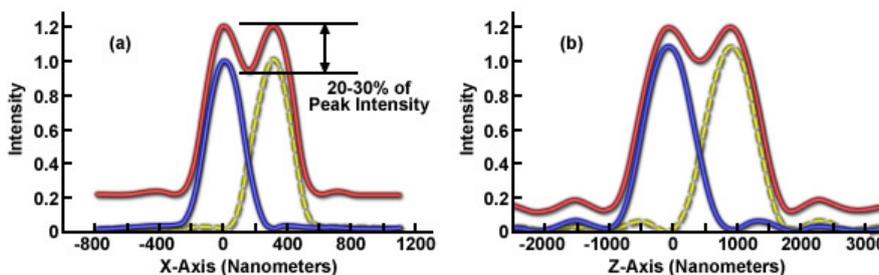
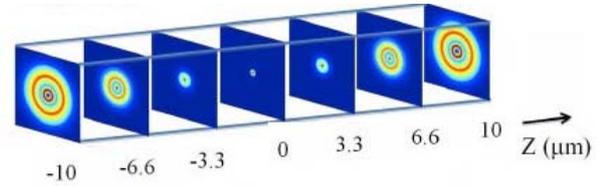


Figure 2

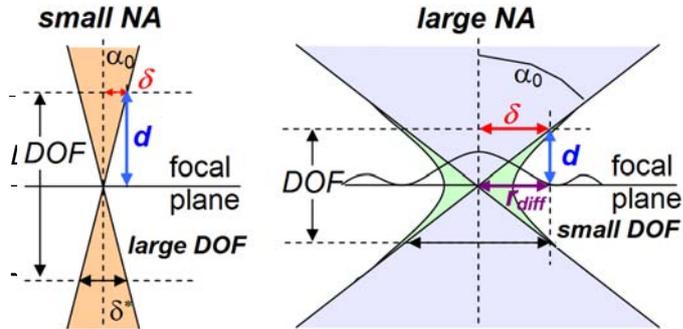
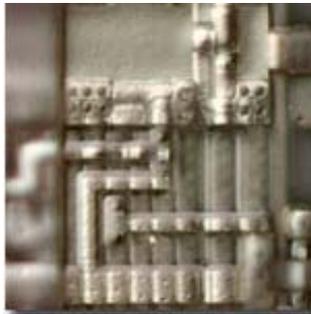
# Depth-of-Field / Focus

Due to the **broadening of the PSF in the axial direction** an object part that is offset from the ideal focal plane appears broadened in the image plane.



The **Depth-of-field (DOF)** is thus defined as the vertical distance or sample height over which the microscopy image appears still sharp.

⇒ Like the lateral resolution, the DOF is determined by acceptance angle  $\alpha_0$  (NA) of the objective lens.



**Derivation of the DOF** using the following resolution criterion:

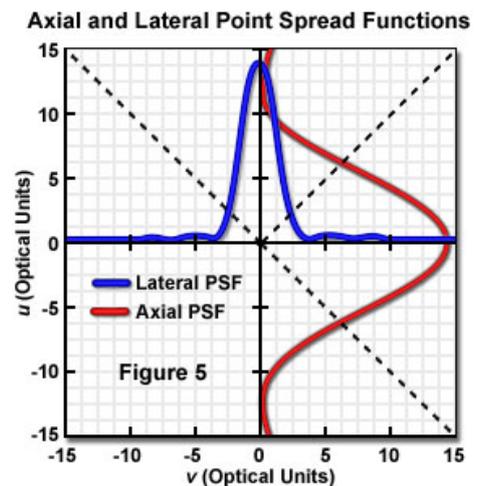
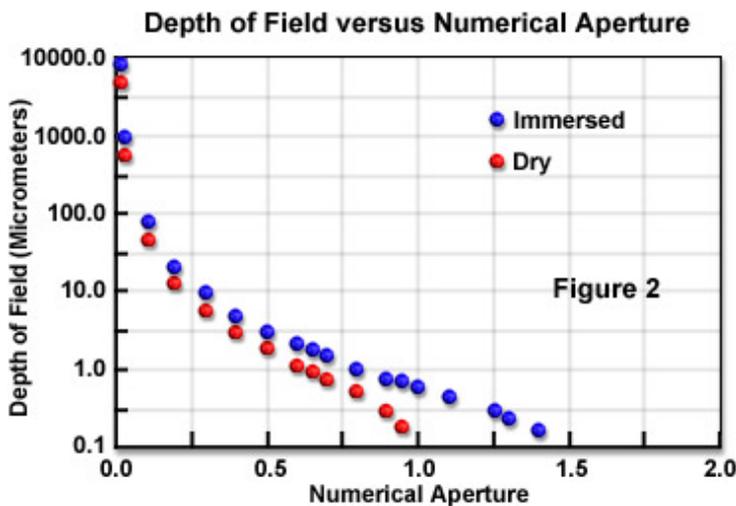
**Criterion:** DOF = Depth of sample across which the **broadening  $\delta(z)$**  due to the spreading of the light cone by  $\alpha_0$  is equal to the lateral resolution  $r_{diff}$  in the focal plane, i.e.,  $\delta(z) = r_{diff}$

From above figure:  $\delta = d \cdot \tan \alpha$  &  $r_{diff} = 0.61 \lambda / n \sin \alpha \Rightarrow d = 0.61 \lambda / (\tan \alpha \cdot n \sin \alpha)$

⇒  $DOF = 2 \cdot d = 1.22 \lambda \cdot (n^2 - NA^2)^{1/2} / NA^2$  For small  $\alpha$ :  $DOF \approx n \cdot \lambda / NA^2$

» Increasing the numerical aperture NA **decreases** the **depth-of-field of the image** !

$DOF = 2 \cdot d = 1.22 \lambda \cdot (n^2 - NA^2)^{1/2} / NA^2$  For small  $\alpha$ :  $DOF \approx n \cdot \lambda / NA^2$



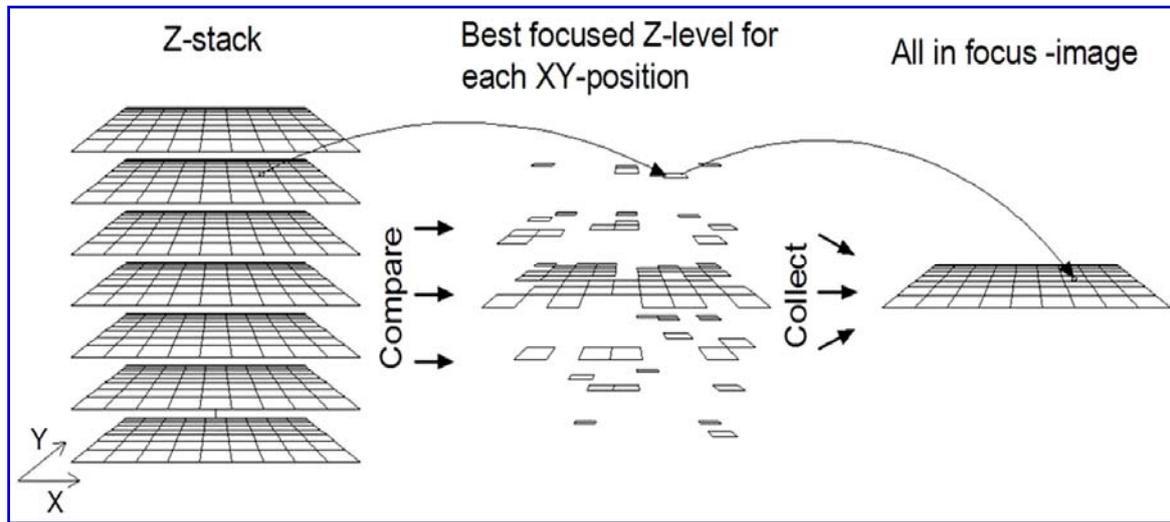
## Conclusions:

- ⇒ For large sample thicknesses or large topography variations, the image is not sharp over the whole microscopy image.
- ⇒ A **high lateral resolution** and a **large DOF cannot be achieved at the same time**.
- ⇒ A **large DOF** mean a **low axial resolution** because at large DOF the microscope cannot discriminate between objects located at different z-positions. This ability is called axial resolution, which is important for 3D imaging of the objects in x,y,z direction.

# 4.9 Z-Series: A Method to Overcome the DOF Limitation

The limit of image sharpness due to the finite DOF can be overcome by recording a whole series of images (= "z-stack") with incremental vertical shifts of the focal plane.

From each image, the **sharp parts** are selected by an image processing software, and then a total **virtual image** is constructed only from the sharp regions for the whole sample series.



### Selection criterion:

The **sharpest z-element** is the one with the highest local contrast modulation, i.e., the element containing the highest spatial frequencies.

## Example: Z-Series in Optical Imaging Microscopy

When a z-image series is recorded with precisely-controlled z-displacement increments  $\Delta z$ , the x,y coordinates of the sharp parts of the images can be correlated with their location in z-direction. In this way, a **complete 3D sample image** is obtained.

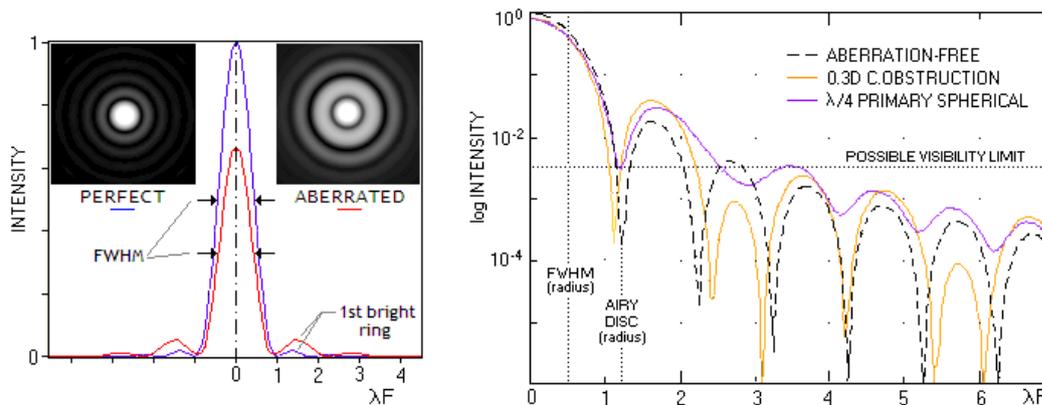
### Examples for z-series 3D image reconstruction

**Lateral resolution:**  $\Delta x = 0.66 \lambda / NA \sim \lambda / 2$     **Vertical resolution:**  $\Delta z = DOF \sim n \cdot \lambda / NA^2 \sim \lambda$

# 4.10 Resolution Limits due to Lens Aberrations

For **imperfect optical systems**, the PSF point spread function is **additionally broadened**.

This is due to the fact that **imperfect lenses** do not focus all beams exactly on the same image spot in the image. This leads to a smearing of the image and thus, the resolution is decreased.



The resulting **effective resolution** is described by the **RMS sum**

of diffraction broadening  $r_{diff}$  **and** aberration broadening  $r_{aber}$  :

$$r_{eff} = (r_{diff}^2 + r_{aber}^2)^{1/2}$$

The overall performance of an objective lens is then

characterized by an **effective numerical aperture  $NA_{eff}$**  :

$$NA_{eff} = n \sin \alpha \cdot r_{diff} / r_{eff}$$

which is always smaller than the geometrical numerical aperture !

There are **three main types of lens errors** (aberrations) that degrade the spatial resolution:

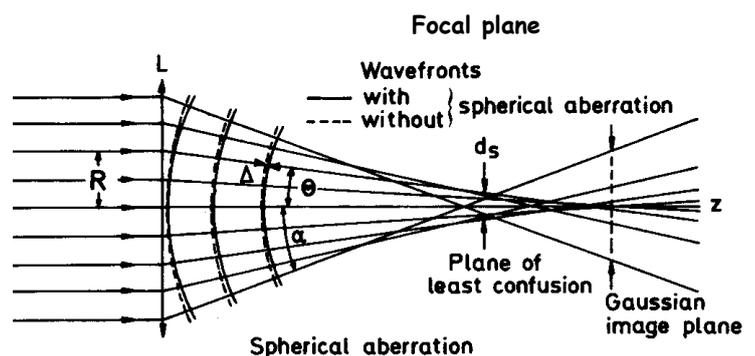
- (1) Spherical Aberration, (2) Chromatic Aberration and (3) Astigmatism.

## 4.10.1 Spherical Aberration

Spherical aberration arises from the **spherical shape of optical lenses**.

A spherically shaped lens acts differently on central and far off-axis rays:

- » Off-axis beams are bend more strongly towards the optical axis.
- » Thus, parallel beams are not focused on the same point on the optical axis.



**Result:** Broadening of the spot size.

**Spot radius  $r_{sph}$  in the focal plane:**  $r_{sph} = C_s \cdot \alpha^n$

$C_s$  ... spherical aberration constant.  
 $\alpha$  - exponent:  $n \sim 3$

**Radius in the plane of least confusion:**  $r_{sph}^* = r_{sph} / 4$

Spherical aberration decreases with *decreasing* lens curvature, i.e., *increasing* focal length.

## Example: Calculation of spherical aberration for a simple *half-spherical* lens

Focal length  $f$  for a thin spherical lens:  $\frac{1}{f} = (n-1) \left( \frac{1}{R_1} - \frac{1}{R_2} \right)$ . Half-spherical lens:  $f_0 = \frac{R}{n-1}$

(1) **Incidence angle**  $\gamma$  on curved lens surface:  $\sin \gamma = y/R$

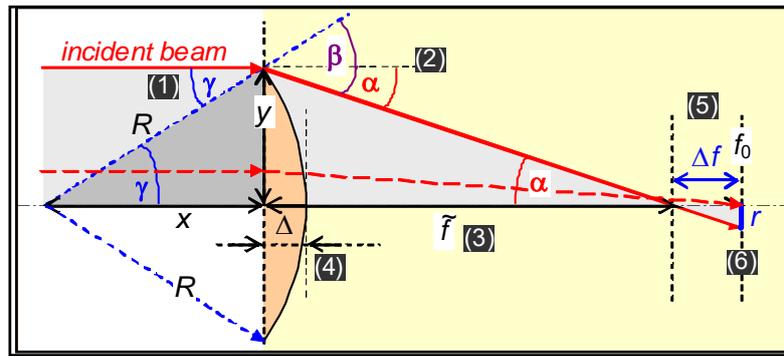
(2) **Refraction** of beam at curved surface (Snell's law):  $\sin \beta = n \cdot \sin \gamma \gg$  Focusing angle  $\alpha = (\beta - \gamma)$

(3) **Focal distance of outer rays**:  $\tilde{f}(y) = y / \tan \alpha \approx f_0 \left[ 1 - k(y/R)^2 \right]$  for  $\alpha < 10^\circ$ ,  $k = \text{constant}$

(4) **Additional shift**  $\Delta$  of origin:  $\Delta(y) = R - x = R(1 - \cos \gamma) \approx 1/2R \cdot (y/R)^2$  for  $\alpha < 10^\circ$

⇒ **Effective focal distance of outer rays**:

$$f(y) = \tilde{f}(y) - \Delta(y)$$



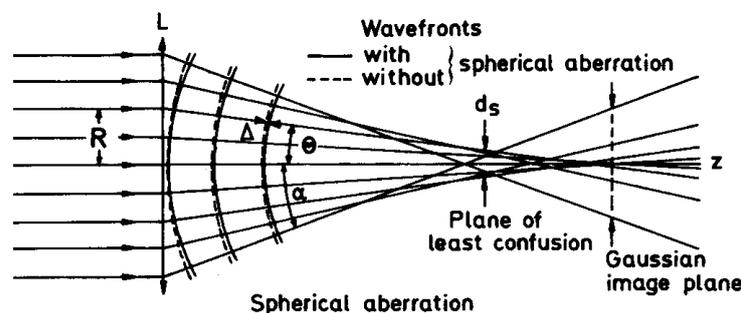
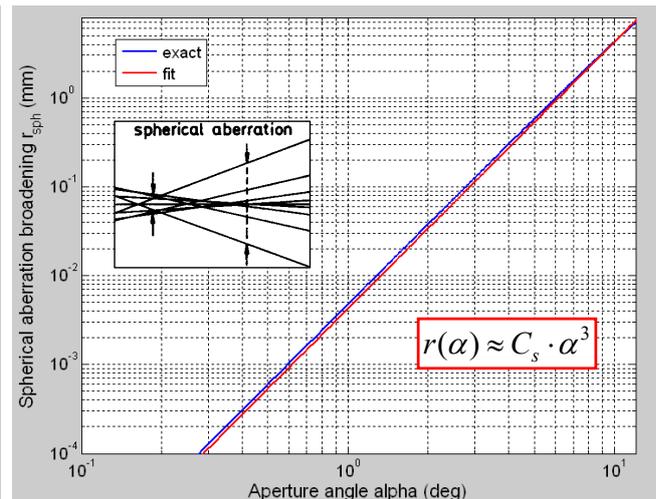
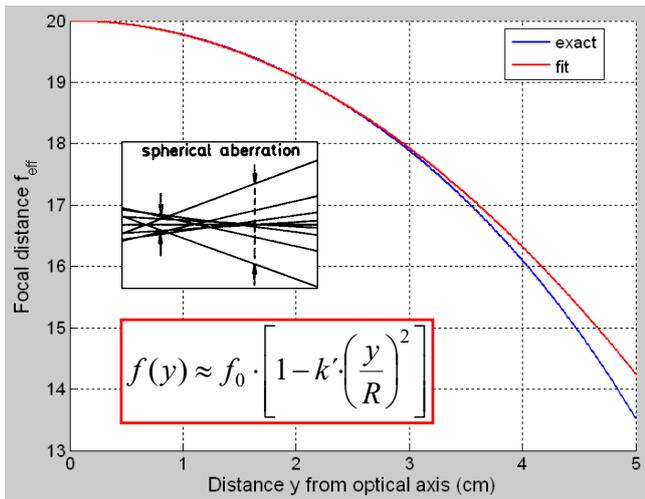
(5) **Change of focal distance**:  $\Delta f(y) = f_0 - \tilde{f} - \Delta \approx f_0 \cdot k' \cdot \left( \frac{y}{R} \right)^2 = f_0(n-1)^2 \cdot k' \cdot \left( \frac{y}{f_0} \right)^2 \approx f_0(n-1)^2 \cdot k' \cdot \alpha^2$

(6) **Broadening**:  $r(\alpha) = \Delta f(\alpha) \cdot \tan \alpha \approx f_0(n-1)^2 \cdot k' \cdot \alpha^3$

$$r(\alpha) \approx C_s \cdot \alpha^3$$

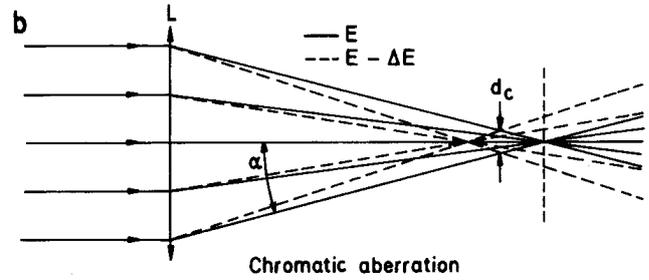
## Focal Distance and Broadening as a Function of Distance and Aperture Angle

**Example:** Curvature  $R = 10$  cm, refractive index  $n = 1.5$ , corresponding focal length  $f_0 = 20$  cm



## 4.10.2 Chromatic Aberration

Light with different wavelengths or energies are refracted differently by the lens due to the **dispersion  $n(\lambda)$  of the refractive index** of the lens material.



- **Broadening of the spot size** (spot radius  $r_{chr}$ ) according to

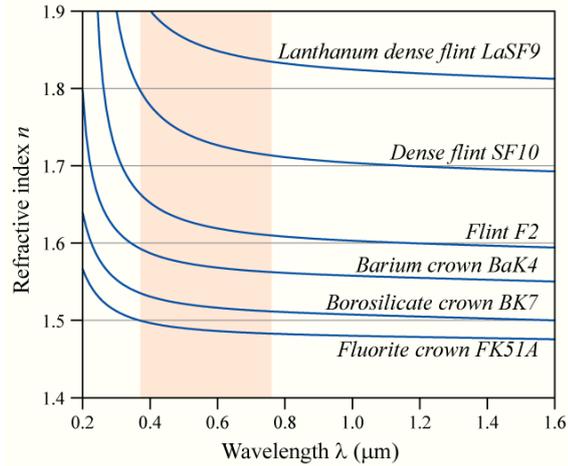
$$\Delta f \sim \Delta \lambda : r_{chr} = C_c \frac{\Delta \lambda}{\lambda} \alpha^m = C_c \frac{\Delta E}{E_0} \alpha^m$$

**$C_c$  ... chromatic aberration constant:**  $\sim dn/d\lambda$  or  $\sim dE/d\lambda$  or  $\sim df/d\lambda$ ,  $m \sim 1$

**Example: Dispersion  $n(\lambda)$  of the refractive index for typical lens**

### Chromatic aberration

- ⇒ strongly depends on the material properties
- ⇒ increases with increasing wavelength spread of the used light



## 4.10.3 Astigmatism

= **Non-uniform action of the lens in two orthogonal directions.**

### Main origin:

**Asymmetries** of the lenses, misalignments of the optical system, electric stray fields due to charging in electron microscopy.

- Broadening of the spot size/radius  $r_{ast}$  according to

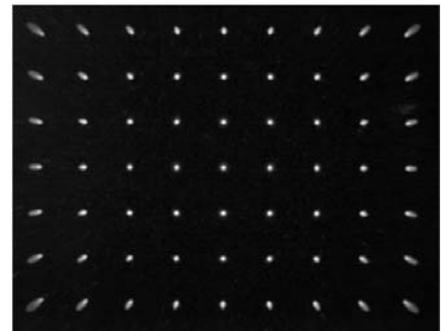
$$r_{ast} = \alpha \Delta f$$

In all, the

- ⇒ **total effective resolution** is given by:

$$r_{eff} = (r_{dif}^2 + r_{sph}^2 + r_{chr}^2 + r_{ast}^2 + \dots)^{1/2}$$

i.e. by the sum of all broadening factors.



# 4.11 Maximizing the Microscope Resolution

For any microscope, the **total effective resolution**  $r_{tot}$  is given by the **sum** of **diffraction and aberration broadening** according to:

**Total effective resolution:** 
$$r_{tot} = \sqrt{r_{dif}^2 + r_{sph}^2 + r_{chr}^2 + r_{ast}^2} = 0.61 \cdot \lambda / NA_{eff}$$

which is characterized by the *effective numerical aperture*  $NA_{eff}$

⇒ For highest resolution **all terms must be minimized!**

However: Diffraction broadening and aberration broadening **counteract** each other!

$$r_{tot} = \sqrt{\left(\frac{0.61\lambda}{n \sin \alpha}\right)^2 + \left(C_{sph} \alpha^3\right)^2 + \left(C_{chr} \frac{\Delta\lambda}{\lambda} \alpha\right)^2 \dots}$$

- Diffraction broadening is proportional to  $\sim 1 / \sin \alpha$

- Aberration broadening is proportional to  $\sim C \cdot \alpha^n$

⇒ With increasing aperture  $\alpha$ , diffraction broadening decreases but lens aberration increase.

## Consequences:

- ⇒ The resolution cannot be simply increased by increasing the lens diameter: Instead a **compromise** must be made that **minimizes the sum** of all broadening factors.
- ⇒ This means that there exists an **optimum aperture angle**  $\alpha_{opt}$  for which the **highest resolution** is practically achieved.
- ⇒ This optimization is particularly important for **electron microscopy**, where the lens aberrations are very large and dominate the achievable resolution.

## 4.11.1 Reducing Diffraction Broadening

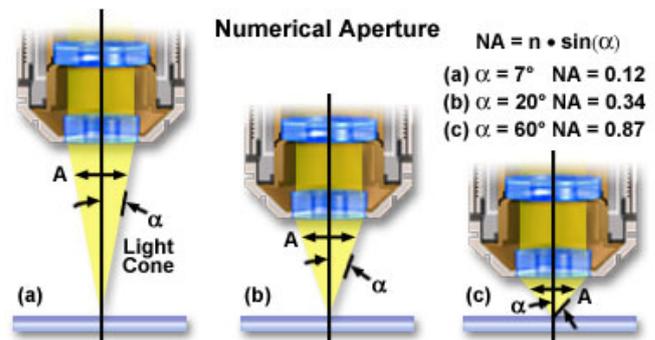
Numerical apertures  $NA = n \cdot \sin \alpha$  : Means that  $\alpha$  as well as  $n$  should be maximized

### (a) Increasing the aperture angle

$$\alpha = \tan(D/2f)$$

requires lenses with **large diameter**  $D$  and/or **smaller focal length**  $f$ .

- ⇒ For microscopes with fixed tube length and with fixed objective diameter  $D$ ,  $\alpha$  can be increased only by **decreasing the focal length**  $f$  of the objective,
- ⇒ Reducing  $f$  means also **reducing the working distance**  $w \sim f$



$$r_{diff} = 0.61 \cdot \lambda / n \sin \alpha$$

**Consequence:** Because at the same time the **magnification**  $M = t/f$ , increasing  $f$  simultaneously increases the magnification:

- ⇒ High NA (and high resolution) is obtained only at high magnification!
- ⇒ A high NA also gives a high **brightness** of the image, which is proportional to  $\sqrt{(NA)}$

Resolution & Numerical Aperture for  $\lambda = 550\text{nm}$  (green)

Magnification	N.A.	Resolution ( $\mu\text{m}$ )
4x	0.10	1.375
10x	0.25	0.61
20x	0.40	0.37
40x	0.65	0.29
60x	0.75	0.29

N.A. = Numerical Aperture

## (b) Increasing the refractive index $n$ : Immersion Microscopy

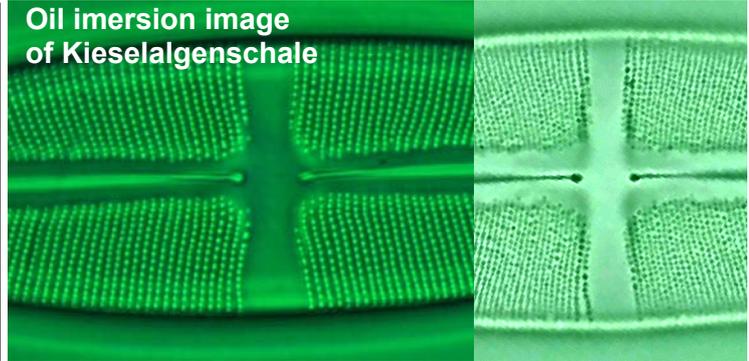
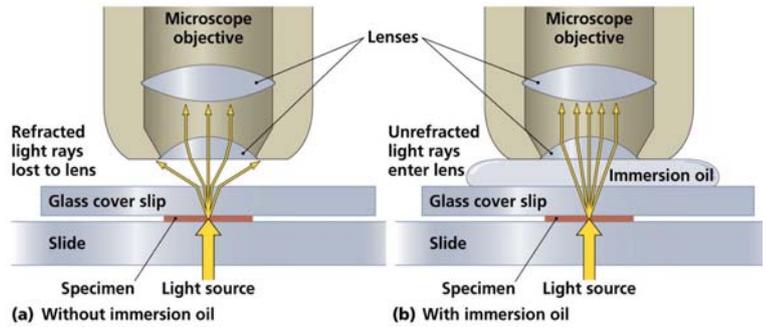
Introducing a **liquid medium** with refractive index  $n > 1$  between the sample and objective reduces the diffraction angles and thus, increases  $\alpha_{eff}$ .

This is called **immersion microscopy**.

Numerical apertures  $NA > 1$  can be obtained, which reduces diffraction broadening and improves resolution

$$r_{diff} = 0.61 \cdot \lambda / n \sin \alpha$$

**air:**  $n = 1$  , **water:**  $n = 1.33$  , **glycerine:**  $n = 1.47$   
**synthetic oils:**  $n = 1.51$ , **Bromonaphthalene:**  $n = 1.659$ , **Methylene iodide:**  $n = 1.74$ ,



### 4.11.2 Correction of Lens Aberrations

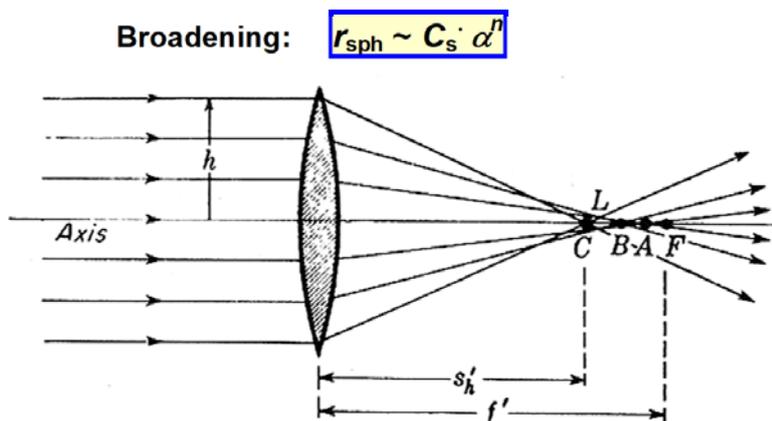
$$r_{eff} = \sqrt{r_{diff}^2 + r_{aber}^2} = 0.61 \cdot \lambda / NA_{eff}$$

⇒ To achieve a diffraction limited resolution, all lens errors must be corrected !

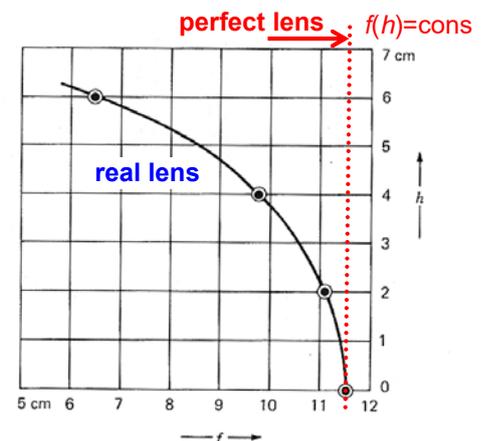
**Non-perfect lens system:** Performance defined by **effective numerical aperture**  $NA_{eff} < n \sin \alpha$ . This value is **smaller** than  $n \sin \alpha$  and accounts for additional broadenings caused by lens aberrations.

- ⇒ Objectives with better corrections yield higher **effective**  $NA_{eff}$  values close to  $NA_{ideal} = n \sin \alpha$
- ⇒ Combination of several lenses in one objective allows to compensate of most lens aberrations.

**(i) Spherical aberration correction:** **Cause:** Spherical curvature of lenses results in stronger bending of outer rays towards the optical axis. Strong effect for "thick" lenses



see: Jenkins & White: Fundamental of Optics



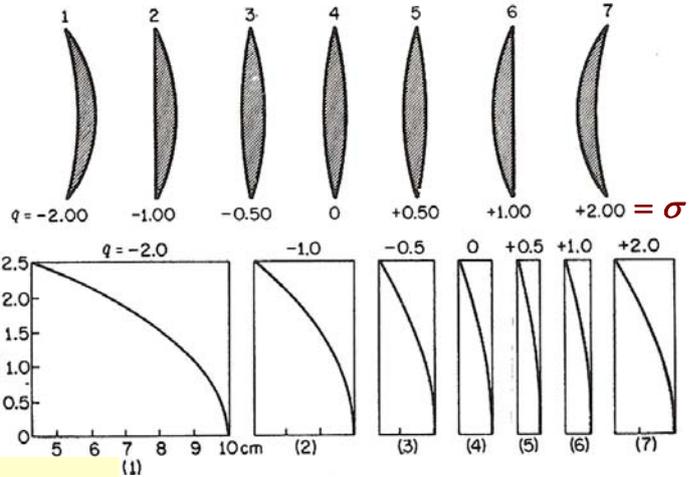
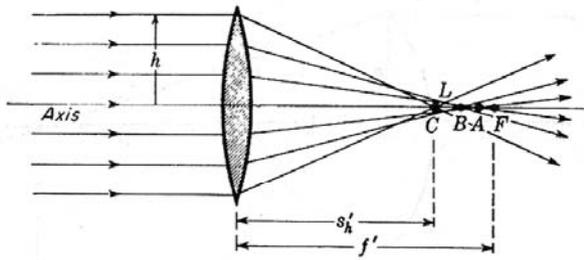
Focal point  $f$  versus distance  $h$  of ray from the optical axis

**Solution #1:** Use of several lenses with smaller curvature and smaller aberration

**Solution #2:** Use of asymmetric lenses with two different curvatures on each side to minimize spherical aberration

Lens shape factor: 
$$\sigma = \frac{R_2 + R_1}{R_2 - R_1}$$

determines the change of  $f(h) \sim$  spherical aberration



Spherical aberration is **minimized** when:

$$\sigma = -\frac{2(n_{lens}^2 - 1)}{n_{lens} + 2} \cdot \left( \frac{s_{obj} - s_{im}}{s_{obj} + s_{im}} \right) = -\frac{2(n_{lens}^2 - 1)}{n_{lens} + 2} \cdot p$$

where  $s_{obj}$  and  $s'_{im}$  are the object and image distances, respectively.

$$p = \left( \frac{s_{object} - s_{image}}{s_{object} + s_{image}} \right)$$

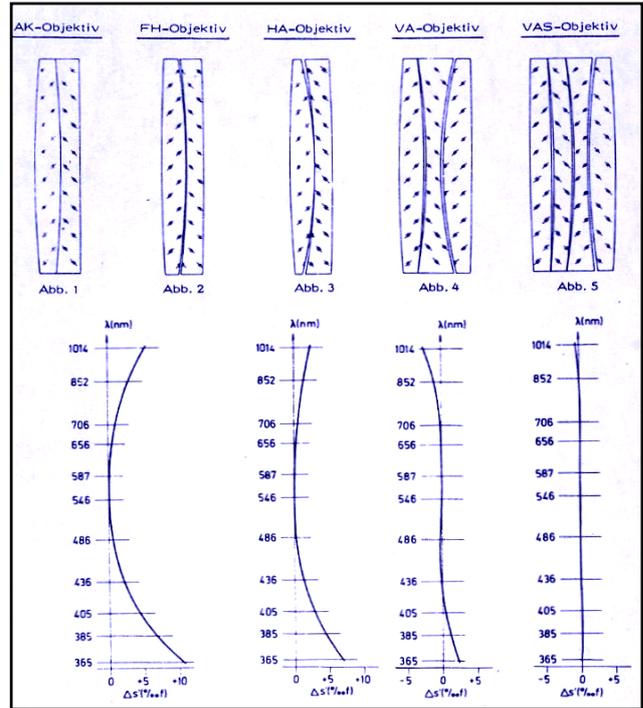
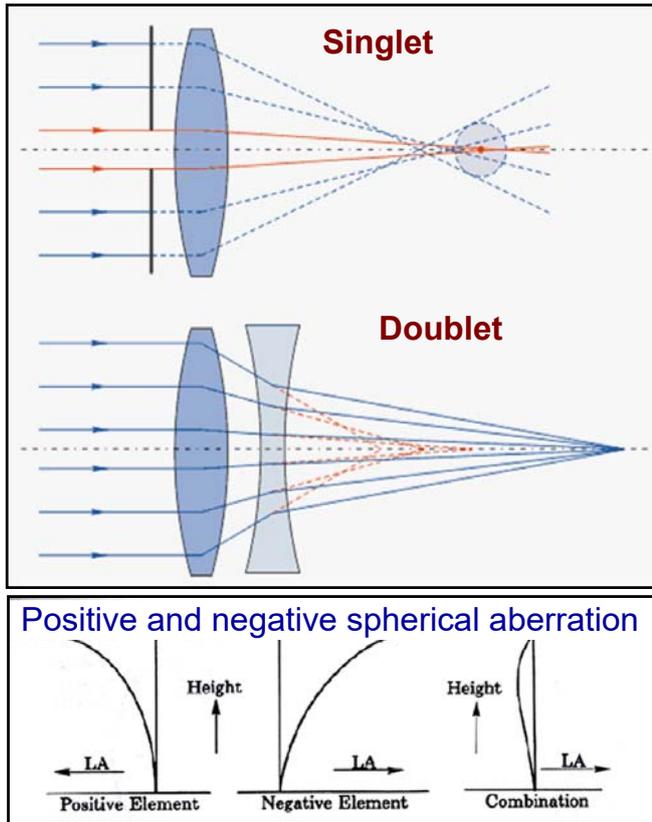
For an object at infinite ( $s_{obj} = \infty, p = -1, n_L = 1.50$ ),

**Optimum lens shape:**

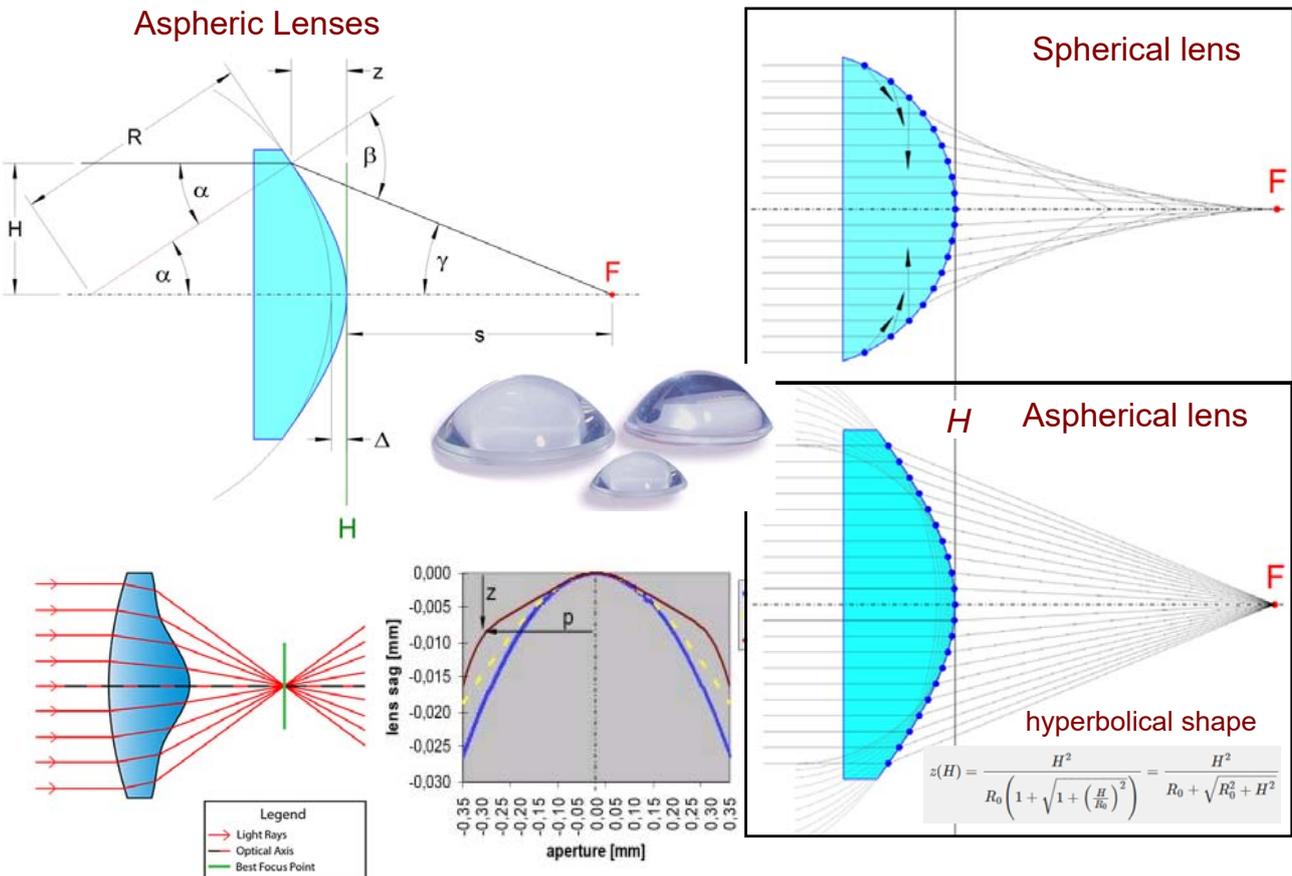
yields

$$\sigma_{opt} \sim 0.7$$

**Solution #3:** Combination of positive (convex) and negative (concave) lenses with **positive** and **negative** spherical aberration constants (**dublets** and triplets)



**Solution #4:** Use of more expensive **nonspherical lenses**



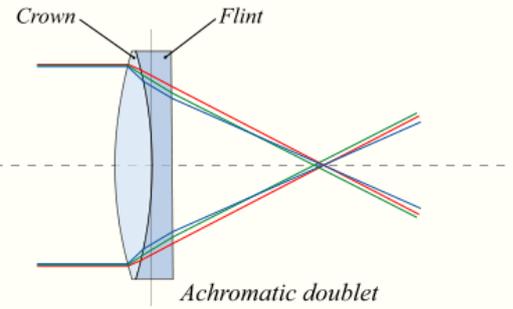
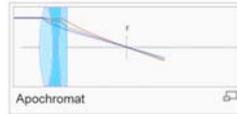
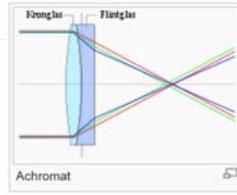
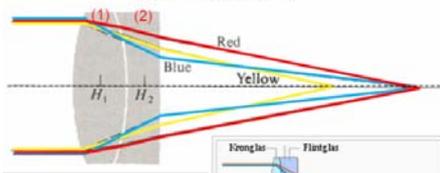
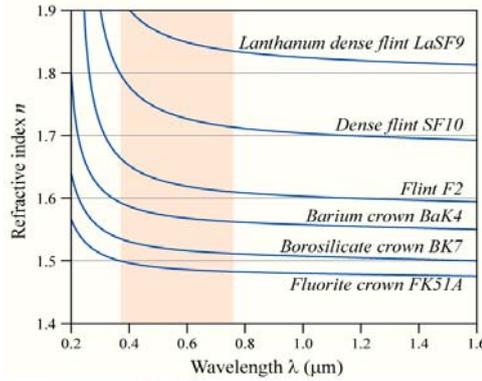
## (ii) Correction of chromatic aberration

### Cause:

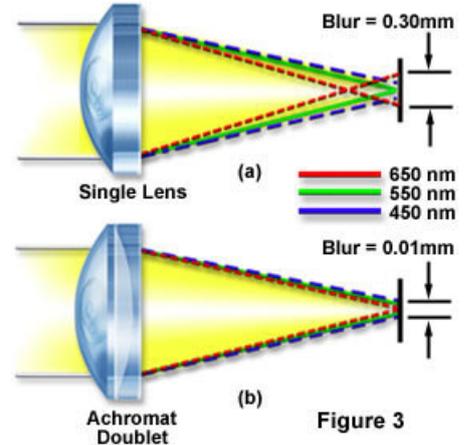
Wavelength dependence, i.e., dispersion of the refractive index  $n(\lambda)$

### Solution:

Lens **doublets** or **triplets** of convex and concave lenses consisting of glass materials with different dispersion (crown, flint glasses and fluorspar).



### Axial Chromatic Aberration



## (iii) Correction of astigmatism

**Cause:** Asymmetric lenses or misaligned/unparallel optical elements

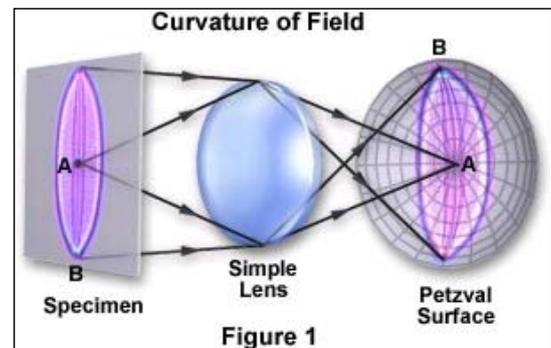
**Solution:** Improved lens symmetry, precise alignment.

## (iv) Curvature of field-of-view

For spherical lenses, the image is not focused on a plane but rather on a curved surface.

**Cause:** Curvature of lens surfaces.

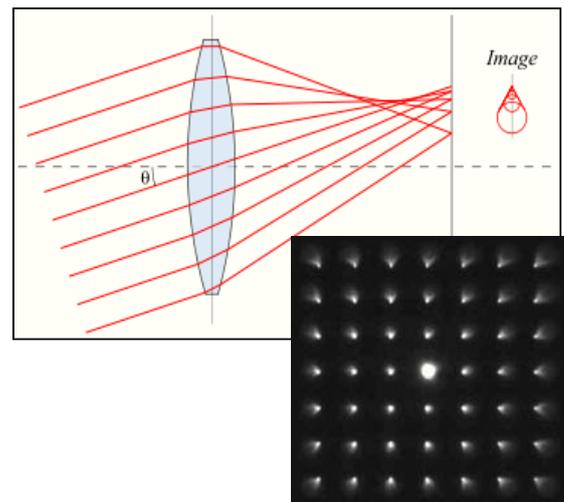
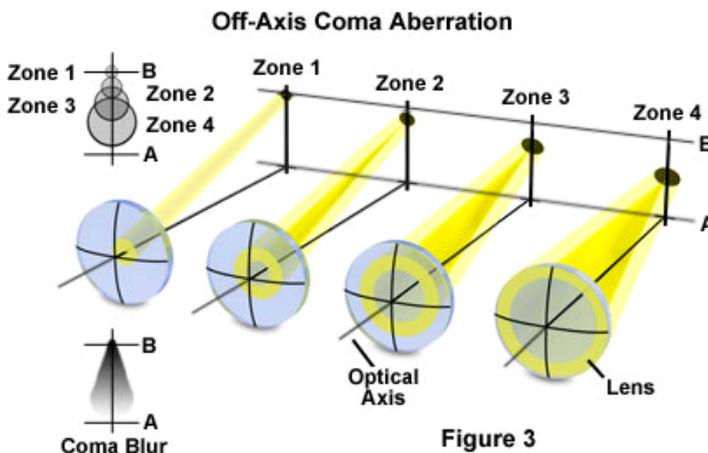
**Solution:** Correction by additional lens elements.



## (v) Coma = smearing of outer part of image

**Cause:** Inclined beams have different focal position for outer beams through the lens.

**Solution:** Correction by additional lens elements.

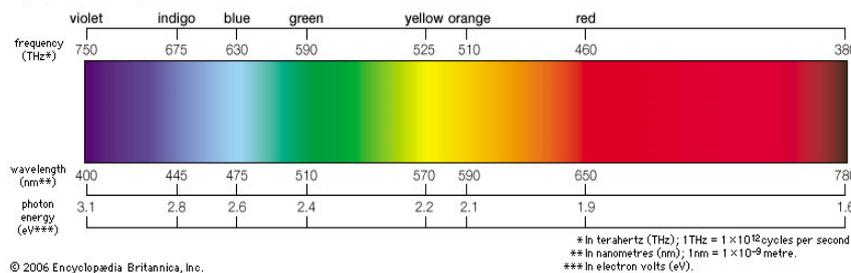


# 4.12 Application to Light Microscopy

In optical microscopy, visible (or near-infrared) light is used for imaging of the sample.

**Wavelength:**  $\lambda[nm] = c / \nu = 1240 / E[eV]$  **Energy:**  $E[eV] = h \cdot \nu = 1240 / \lambda[nm]$

Light, the visible spectrum



Color	Energy	Wavelength
violet	3.2 – 2.8	380–450 nm
blue	2.8 – 2.5	450–495 nm
green	2.5 – 2.2	495–570 nm
yellow	2.2 – 2.1	570–590 nm
orange	2.1 – 1.9	590–620 nm
red	1.9 – 1.6	620–750 nm

**Resolution limits:** Since for VIS & NIR optical systems, effective aberration corrections exist that nearly fully eliminate aberration broadening, **diffraction** is the main remaining **limitation** for the spatial **resolution** of VIS & NIR microscopy.

⇒ The best resolution is about  $\frac{1}{2} \lambda$  of the illumination wavelength  $\lambda$  using aberration corrected objectives with high NA  $\sim 1$

$$r_{res} = r_{diff} = 0.61 \lambda / NA$$

⇒ Note: Recently, sub  $\lambda/2$  resolution scanning fluorescence microscopy has been developed.

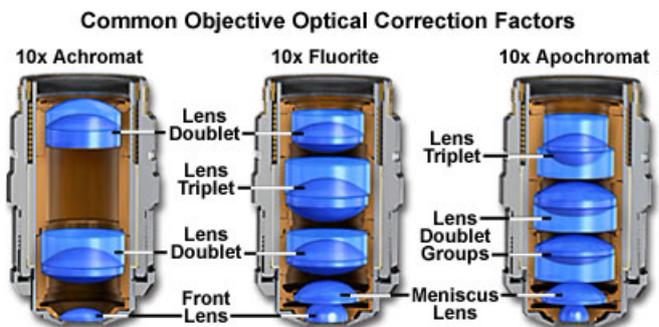
Wavelength (nm)	Resolution (nm)
360 – 450 nm (violet)	190 – 250 nm
450 – 500 nm (blue)	250 – 300 nm
500 – 570 nm (green)	300 – 350 nm
620 – 750 nm (red)	380 – 460 nm

## Types and Specifications of Objective Lenses

**Achromats:** Objectives corrected for axial chromatic aberration **at two wavelengths** (486 nm blue and 656 nm red) and corrected for spherical aberration in the color green (546 nm; see table).

**Fluorites or semi-apochromats:** Additional lenses for spherical corrections at two wavelengths.

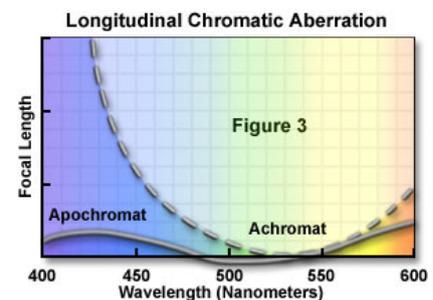
**Apochromats:** Best corrected objectives, correction of spherical aberrations for wide a wavelength region.



### Objective Correction for Optical Aberration

Objective Type	Spherical Aberration	Chromatic Aberration	Field Curvature
Achromat	1 Color	2 Colors	No
Plan Achromat	1 Color	2 Colors	Yes
Fluorite	2-3 Colors	2-3 Colors	No
Plan Fluorite	3-4 Colors	2-4 Colors	Yes
Plan Apochromat	3-4 Colors	4-5 Colors	Yes

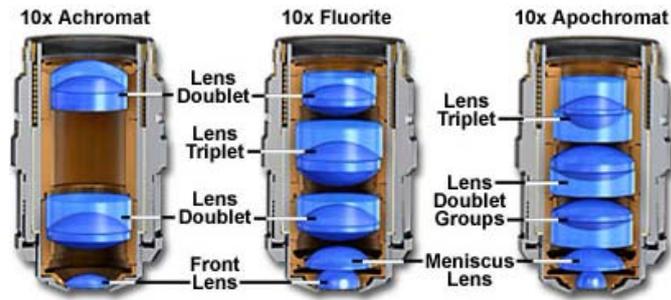
see: [www.micro.magnet.fsu.edu](http://www.micro.magnet.fsu.edu)



**Plan:** Lenses with **field of view curvature correction** are termed with additional “**plan**” prefix.

# Specifications of Objective Lenses

- Type of Objective (Degree of correction)
- Magnification (number)
- Numerical Aperture
- Tube length (mm)
- Cover Glass correction (thickness)



## Resolution and Numerical Aperture for $\lambda = 550\text{nm}$ (green)

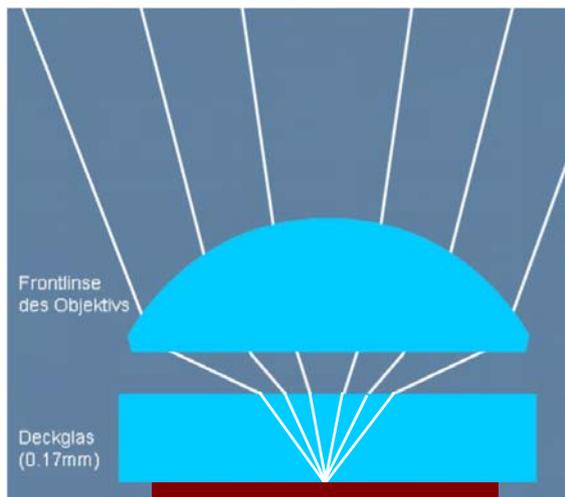
Magnification	Objective Type					
	Plan Achromat		Plan Fluorite		Plan Apochromat	
	N.A.	Resolution ( $\mu\text{m}$ )	N.A.	Resolution ( $\mu\text{m}$ )	N.A.	Resolution ( $\mu\text{m}$ )
4x	0.10	2.75	0.13	2.12	0.20	1.375
10x	0.25	1.10	0.30	0.92	0.45	0.61
20x	0.40	0.69	0.50	0.55	0.75	0.37
40x	0.65	0.42	0.75	0.37	0.95	0.29
60x	0.75	0.37	0.85	0.32	0.95	0.29

N.A. = Numerical Aperture

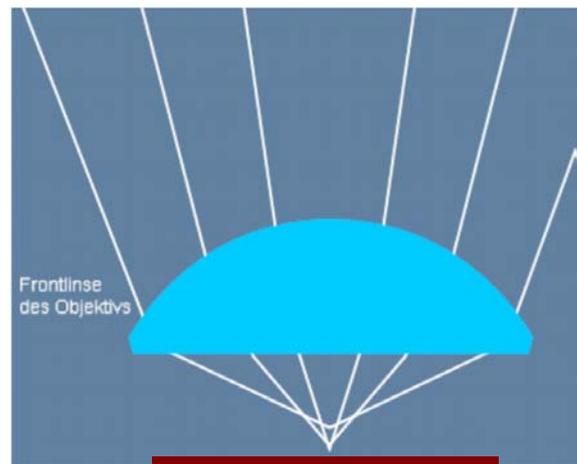
Violet:  $\lambda = 420\text{nm}$  (25% improvement) [www.micro.magnet.fsu.edu](http://www.micro.magnet.fsu.edu)

» Objectives with better corrections yield significantly higher resolution !

## Cover Glass Correction (common for biological specimen)



specimen



specimen

## Comparison of Image Quality: Achrochromat

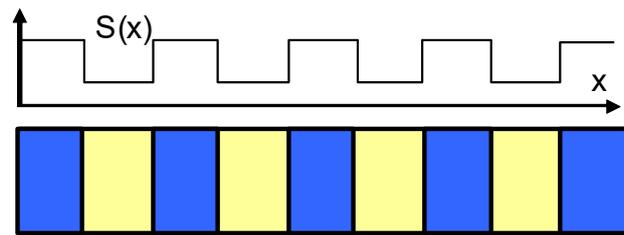
## Planapochromat

## 4.13 Contrast in Microscopy

**Contrast** is produced by local variations in sample properties, which gives rise to changes in the response signal  $S(x,y)$  that is detected and forms the microscopy image.

Contrast is **defined** as **relative difference in signal  $\Delta S$**  from adjacent spots on the sample normalized to the average value:

$$C_{im}(x) = \frac{S(x_1) - S(x_2)}{\bar{S}}$$



**Contrast is crucial** for high resolution microscopy, because if the contrast is smaller than the noise level, the actual resolution is zero ! Thus, contrast needs to be **maximized** for best imaging conditions and **highest resolution**.

**Contrast depends on many factors:**

- ❖ How strongly the local **sample properties** and vary across the sample, i.e., what kind of micro- or nanostructure is present in the sample,
- ❖ The type and strength of the **probe-sample interaction**,
- ❖ The **probe energy or wavelength**,
- ❖ The **type of detected response signal** and what **detection scheme** is used (e.g., spectroscopic detection, angle resolved detection, energy or wavelength filtering, etc. )

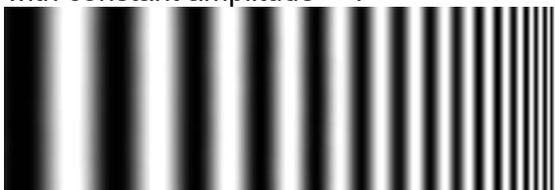
⇒ Contrast depends not only on the sample properties but also on the **imaging conditions** !!

### 4.13.1 Contrast and Resolution

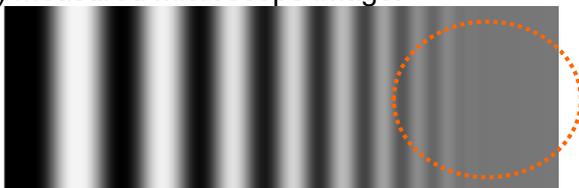
**Contrast has a significant effect on resolution.** This is because small sample features can be **resolved only** when the **signal difference  $\Delta S$  in the image is larger than a certain threshold value** (=visibility criterion). Due to the **broadening of the image** by diffraction or lens aberrations, when the feature size decreases the signal difference  $\Delta S$  or signal amplitude **continuously decreases** and drops to zero when the feature size becomes smaller than the resolution limit.

**Example:** Stripes with decreasing line spacing

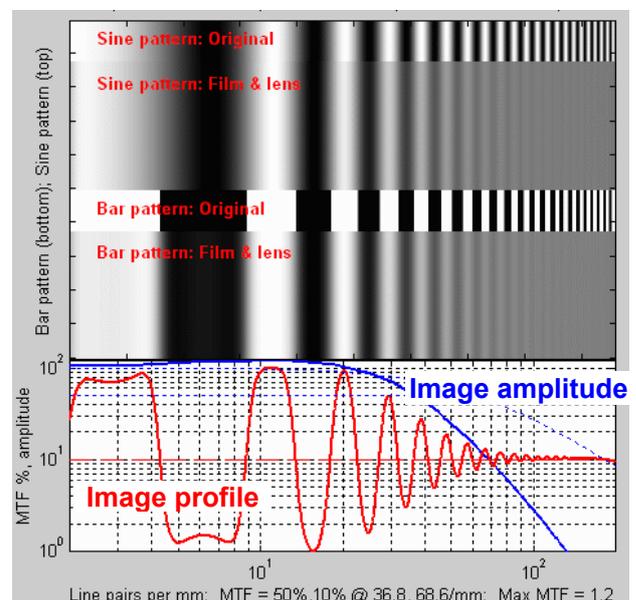
(a) Object of sinusoidal contrast line grating with constant amplitude = 1



(b) Measured microscope image:



⇒ The image contrast decreases when the line spacing decreases !



## 4.13.2 Visibility and Noise in Practical Microscopy Resolution

Visibility defines how well different features in an image can be distinguished.

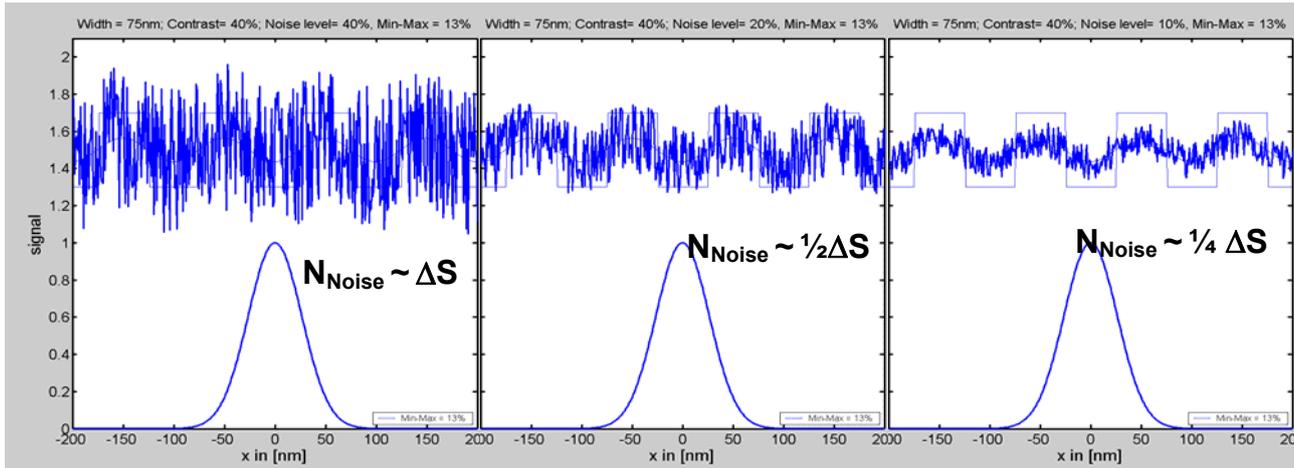
**Visibility criterion:** Adjacent features in an image can only be discerned if the **signal contrast  $\Delta S$**  is **~ five times larger than the noise level  $N$** .

$$\Delta S > 5\sqrt{n}$$

For random noise, the noise level is typically assumed to be proportional to the root mean square deviation of the signal from its average value, i.e.,  $N = \sigma_{RMS} = \sqrt{n}$

As a result, the **signal-to-noise ratio  $S/N$**  strongly influences the **practically achievable microscope resolution**. This means that the theoretically **resolution limit** can be reached only for (i) **high contrast specimen** and/or (ii) **sufficiently low noise levels** in the images.

**Example:** Image signal  $S(x)$  for a periodic sample  $C(x)$  images with a Gaussian response function for different superimposed **uncorrelated random noise levels  $N$** .



**Sources of noise:** Fluctuations of illumination intensity, of sample emission, detector noise, amplifier noise, mechanical and acoustic noise, etc. ....

Noise is particular a **problem for scanning microscopy** because the measurement time per pixel or image point is very small (= total recording time / number of pixels ( $10^6$ )).

**Noise reduction can be achieved by:**

$$S(t) = \bar{S} + N(t)$$

- » **improved hardware** (low noise detectors, amplifiers, vibration damping, electromagnetic shielding)
- » **increasing the illumination intensity** when the absolute noise level is constant.
- » **time averaging** or signal integration.

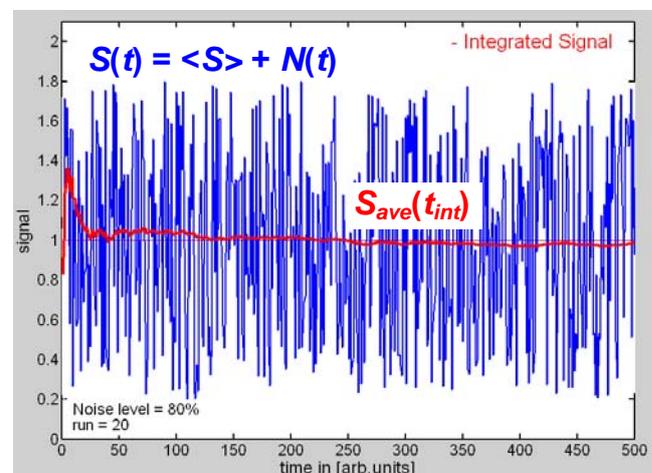
**Time averaging and integration:**

For **uncorrelated statistical noise**, the **root mean square (RMS) noise level  $\sigma$**  decreases with **increasing averaging or integration time**.

This is because uncorrelated random noise averages out, leaving a constant signal value after a sufficiently long integration time.

$$S_{ave}(\tau) = \tau^{-1} \int_0^\tau (\bar{S} + N(t)) dt$$

$$= \bar{S} + \tau^{-1} \int_0^\tau N(t) dt \rightarrow \bar{S}$$

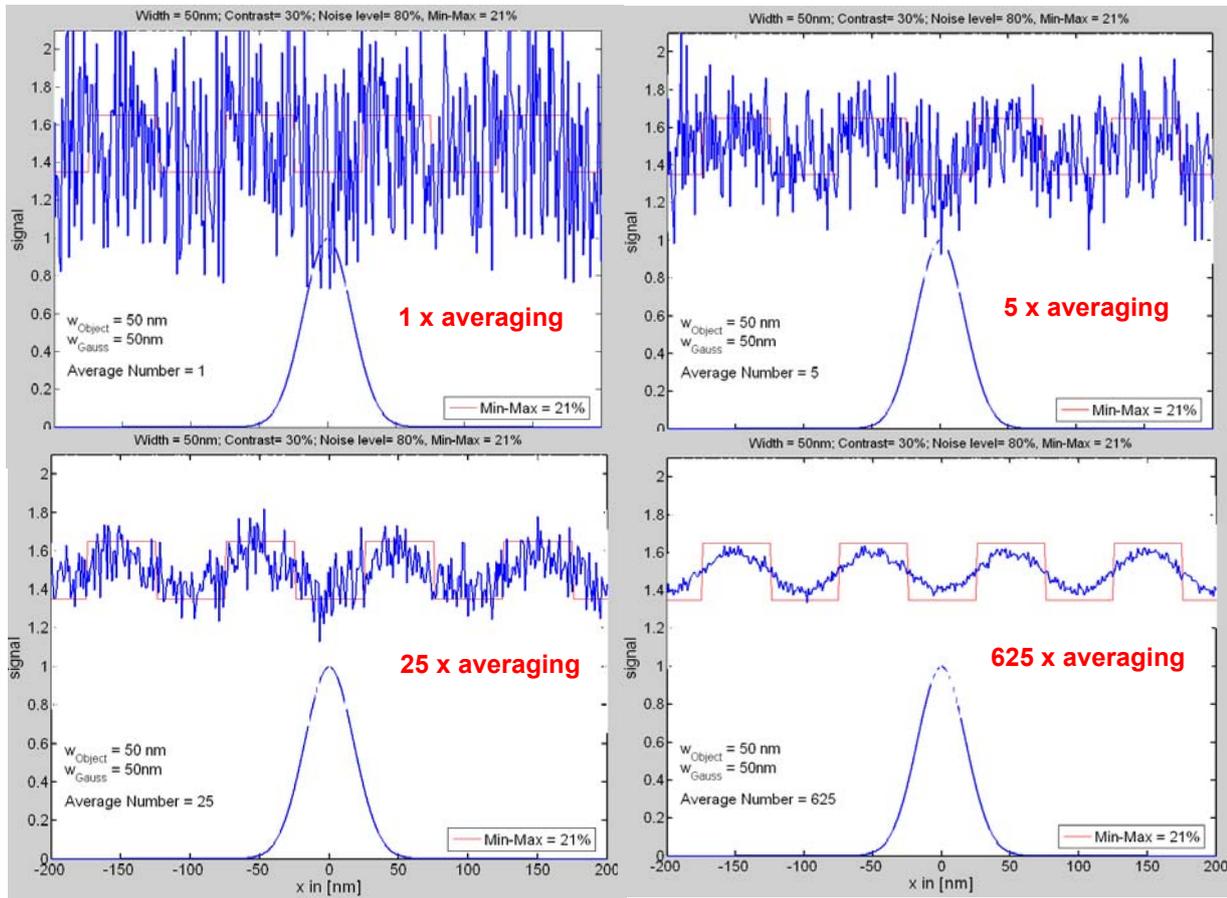


**Resulting noise level  $\sigma_0^{RMS}$**  of average signal:

When the signal is measured  $n$ -times giving a set of values  $S_i$  the RMS noise of the average signal is given by the standard deviation of the  $S_i$  values:

$$\sigma_0^{RMS} = \sqrt{\frac{1}{n} \sum_{i=1}^n (S_i - \bar{S})^2}$$

## Example how the signal-to-noise ratio improves by time averaging:



## Example: Noise Reduction in Scanning Electron Microscopy

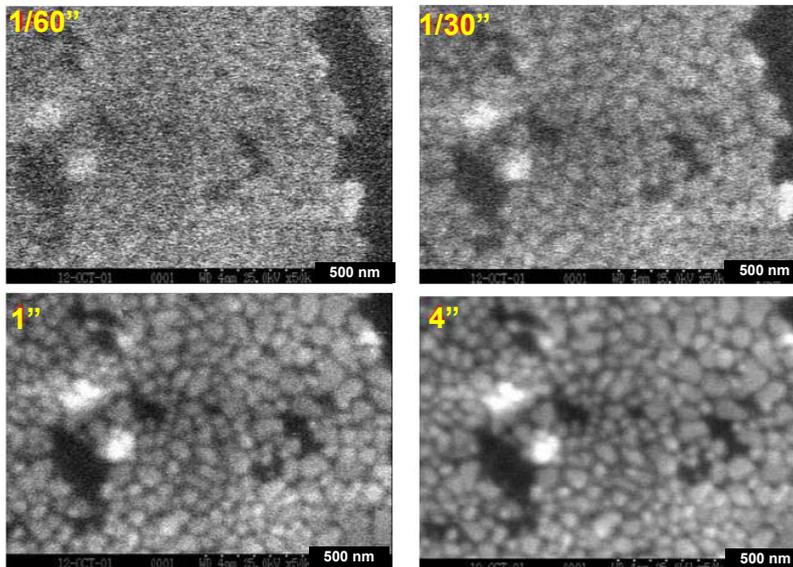
According to statistics, the **RMS noise**  $\sigma_{n,ave}$  of an  $n$ -times averaged signal decreases according to:

$$\sigma_{n,ave} = \sigma / \sqrt{n}$$

This means that the noise  $\sigma_{n,ave}$  of the average signal decreases by the factor  $1/\sqrt{n}$  with increasing number  $n$  of averaging steps, which is equivalent to increasing the integration time  $\tau$ .

Thus, the **signal-to-noise ratio**  $S/N = S/\sigma$  improves as  $S/N = S/\sigma_n = (S/\sigma_0^{RMS})\sqrt{n}$

### Example: SEM images obtained using different integration times



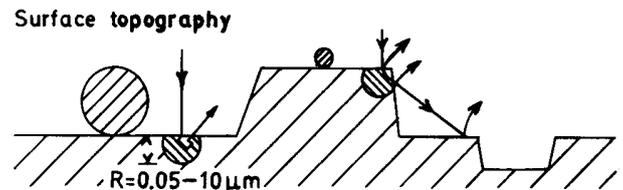
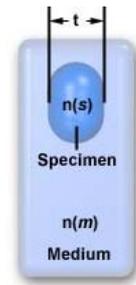
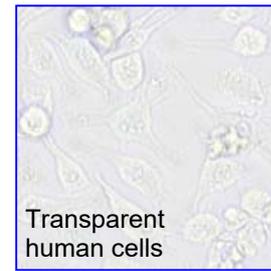
SEM images of 50-200 nm Au crystallites on a carbon template recorded with different integration times (in seconds).

$$\frac{\text{Signal}}{\text{Noise}} = \frac{S}{N} = \frac{\bar{n}}{\sqrt{\bar{n}}} = \sqrt{\bar{n}}$$

## 4.14 Contrast Mechanisms

Generally, **many different mechanisms** contribute to the contrast in the response signal:

- ❖ **Absorption and thickness contrast** due to variation of the transmission according to  $T(x) \sim e^{-\alpha(x)d(x)}$
- ❖ **Reflection/backscattering contrast** of the probe due to variation of the reflectivity and backscattering yield
- ❖ **Phase/interference contrast** due to constructive or destructive interference of beams experiencing different phase shifts when transmitted or reflected from the sample,
- ❖ **Topographic contrast** due to local tilts, edges, shadowing, multiple scattering, etc.,
- ❖ **Diffraction & angular scattering contrast** due to sideways deflection by diffraction or by diffuse scattering at surface roughness, topographic features or local microstructure,
- ❖ **Spectroscopic / color contrast** due to energy/wavelength differences in the transmitted beam or in the excited response signal such as, e.g., fluorescence, luminescence, cathodoluminescence, characteristic x-ray emission, fluorescence markers, etc. (see Chap. 2).



**Contrast formation** is often **very complex**. This applies in particular for electron microscopy where many different types of interaction processes occur simultaneously. In addition, the contrast mechanisms can **vary across the sample surface** when the topography is complex.

## Contrast for the Different Microscopy Methods

### Optical microscopy:

- Contrast due to local difference in **optical constants**  $n(x)$  and  $\alpha(x)$ , color contrast,
- Contrast in **fluorescence yield**, **topographic contrast** (reflection mode), ..

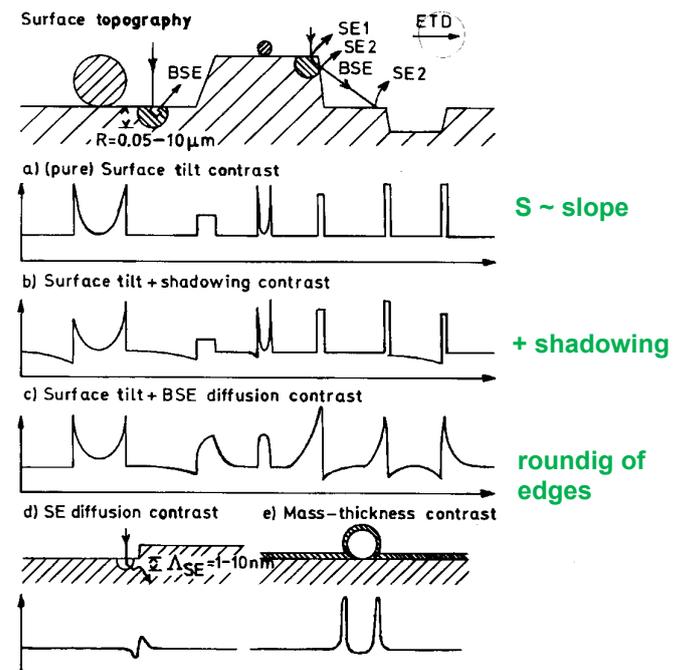
### X-rays:

- **Z-contrast** due to different x-ray absorption.
- Strong **diffraction contrast** due to local variation in lattice structure and lattice constants, strain contrast,

### Electrons:

- **Z-contrast** due to dependence of electron scattering on the atomic Z-number.
- **Absorption contrast** and contrast due to variation of **secondary electron generation yield** and diffuse scattering
- **Topographic contrast** due to shadowing and edge effects (SEM).
- **Diffraction and phase contrast** (TEM) due to differences in the lattice structure and lattice constants.

### Example: Scanning Electron Microscopy



**Fig. 6.1.** Contributions to topographic contrast demonstrated schematically by surface contours (top) and linescans of SE signals; (a) surface tilt contrast, (b) shadowing contrast, (c) BSE diffusion contrast, (d) SE diffusion contrast and (e) mass-thickness contrast

## 4.15 Contrast Enhancement Techniques

For many sample specimen, the relative differences in the transmitted light or for electron microscopy is often rather small and thus, the image **contrast weak**.

In order to fulfill the visibility criterion and actually discern different features, therefore **contrast enhancement techniques** are employed, by which the magnitude of the contrast in the microscopy images is increased.

### Contrast enhancement techniques for imaging (and scanning) microscopy

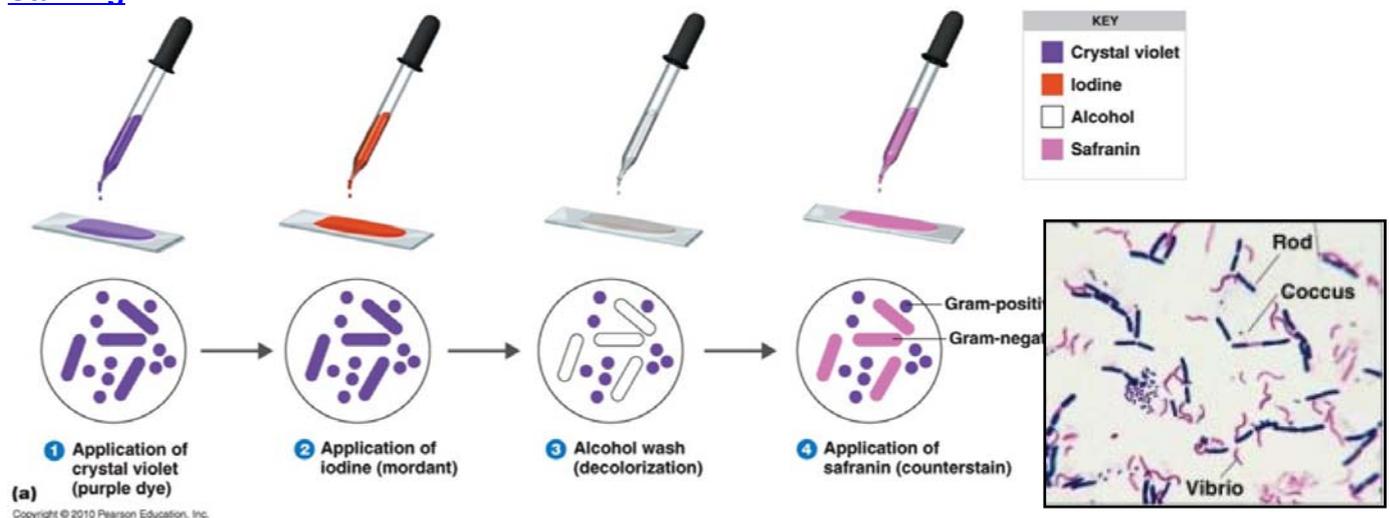
1. **Sample modification**: Staining (dyes), selective etching, fluorescence markers.
2. **Grazing incidence illumination** and/or **detection** (=angular filtering) to enhance topography contrast due to increased shadowing and scattering at step edges.
3. **Spectral filtering** of detected signal to take advantage of differences in  $n(E,\lambda)$ ,  $\alpha(E,\lambda)$ ,  $I(E,\lambda)$  of different parts of the sample and to obtain enhanced material selectivity.
4. **Modification of the optical path** in the microscope using additional beam blocking or phase shifting elements. This results in **dark field microscopy, phase contrast microscopy, differential interference contrast microscopy, etc.**

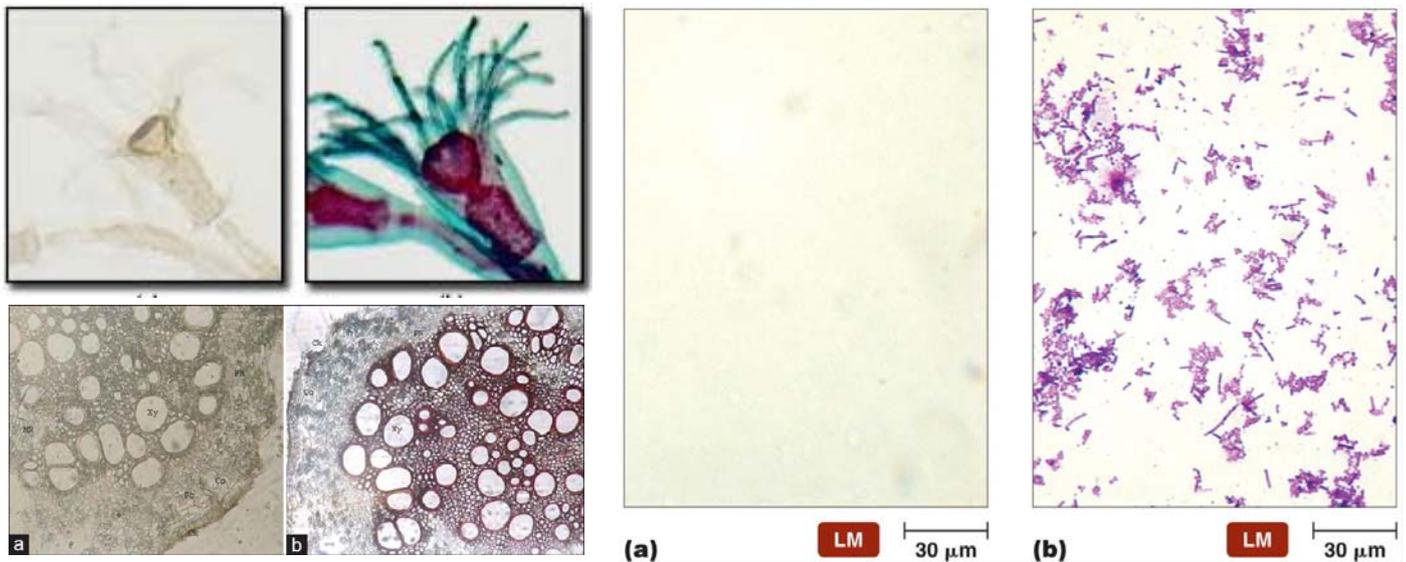
**Note:** In general, the image contrast is not solely determined by the sample properties, but also depends strongly on the light path and optical configuration of the imaging optics which can be modified and optimized for a given application.

### 4.15.1 Staining and Sample Modification

- ⇒ Local **modification of the absorption or emission of biological or organic specimen** by selective absorption of color dye molecules that attach to specific sites of the sample structure.
- ⇒ For **metallurgical specimen** often etching or staining is applied that selectively removes material according to its composition or local orientation, .....

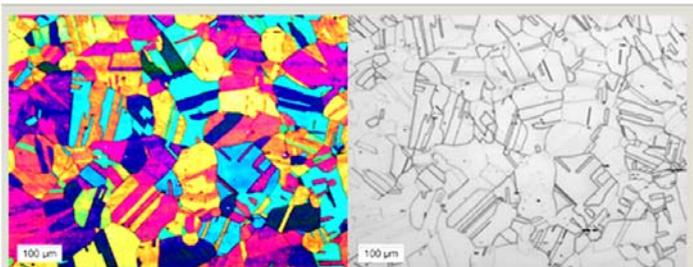
#### Staining:





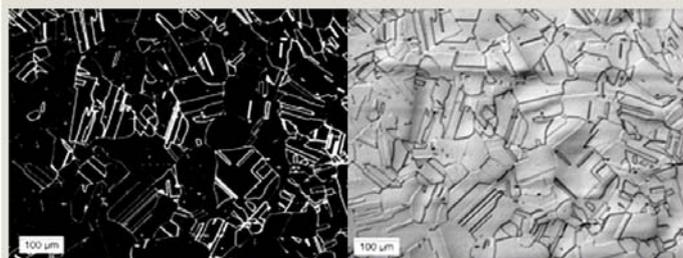
### Example: Etched Metallurgical Specimen

The etchants used in metallographic examination are solutions of acid and chemicals and are applied to attack selectively a polished surface, thus permitting microstructural examination



(a) Stain etchant used to produce color

(b) Bright field – typical way of examining microstructure, shows grain structure and “twins,” but some areas not clearly defined

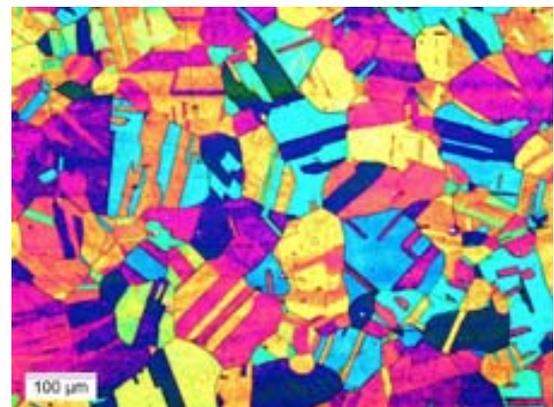


((c) Dark field – shows negative effect with grain boundaries more distinct, but some other areas not totally defined

(d) Three-dimensional effect makes grain boundaries clearer, with more sharply defined grains and “twins.”

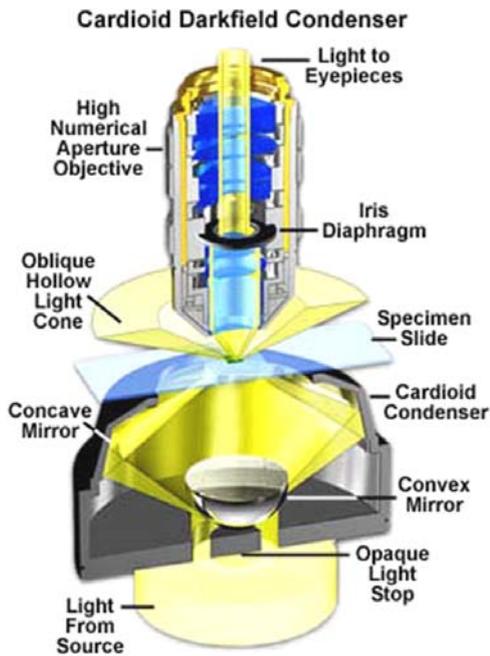
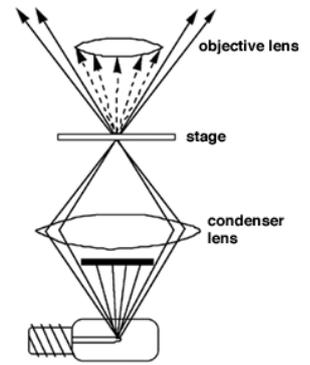
Samples (b), (c) and (d) were etched using Waterless Kalling’s Reagent.

Microstructure of a stainless Type 330 sample @ 100 x magnification, prepared using a tint etch consisting of a solution of 40 ml hydrochloric acid (HCL) + distilled water (H<sub>2</sub>O) + one gram potassium meta bisulfite (K<sub>2</sub>S<sub>2</sub>O<sub>5</sub>) + 4 grams ammonium bifluoride (NH<sub>4</sub>F–HF) at room temperature.

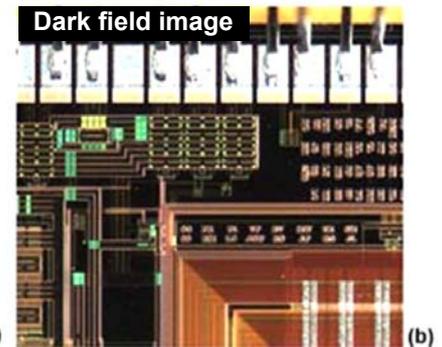
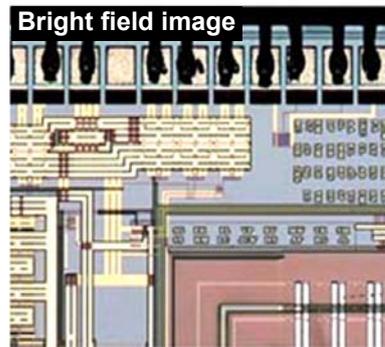


## 4.15.2 Dark-Field Microscopy (=enhances edge contrast)

**Working principle:** *Off-axis illumination* of the sample and **blocking** the central light passing through the specimen without angular scattering or deflection. The **image** is then **formed exclusively by scattered light**. Without scattered light, the image appears dark (=dark-field image).

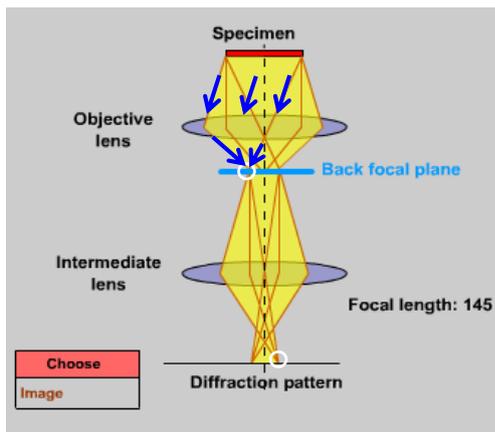


In **dark-field imaging**, the specimen is illuminated with a cone of oblique rays. **Without angular scattering** in the sample these rays pass outside of the objective lens and not contribute to the image. The specimen therefore **appears dark** (= "dark-field microscopy"). Only at features such as steps or interfaces where light is scattered the sample appears bright, i.e., the **contrast is drastically enhanced**. The same principle is used in dark-field transmission electron microscopy.



## Bright-field / Dark-field Imaging in TEM

In TEM, bright-field or dark-field contrast is obtained by blocking part of the beams using apertures inserted in the back focal plane where the diffraction pattern of the sample is formed.



**Dark field:** Central intensity maximum is blocked and only angularly deflected electrons are selected for image formation. Thus, the sample appears dark, except for parts with enhanced angular deflection.

**Bright field:** Only central intensity maximum is used for image formation.

At imperfections, electrons are sideways deflected and thus, they appear dark.

### Example: TEM imaging of a thin Ti foil

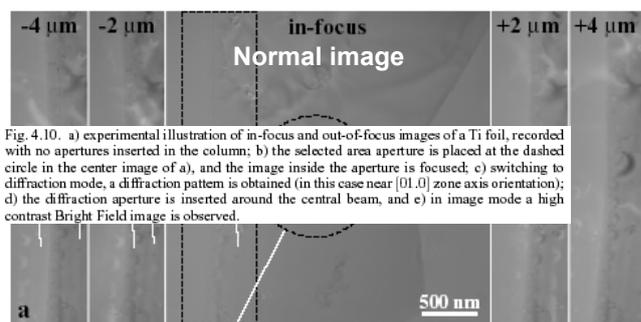
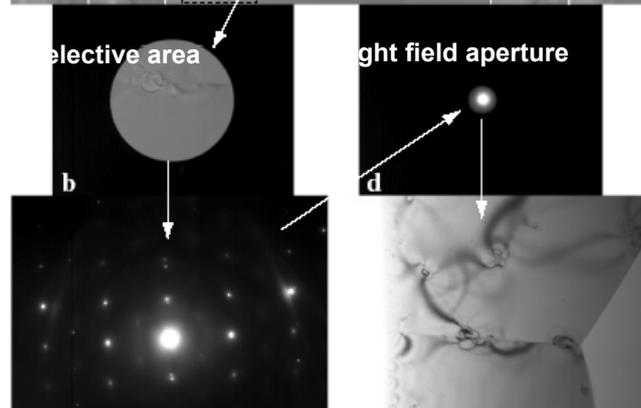


Fig. 4.10. a) experimental illustration of in-focus and out-of-focus images of a Ti foil, recorded with no apertures inserted in the column; b) the selected area aperture is placed at the dashed circle in the center image of a), and the image inside the aperture is focused; c) switching to diffraction mode, a diffraction pattern is obtained (in this case near  $[01.0]$  zone axis orientation); d) the diffraction aperture is inserted around the central beam, and e) in image mode a high contrast Bright Field image is observed.

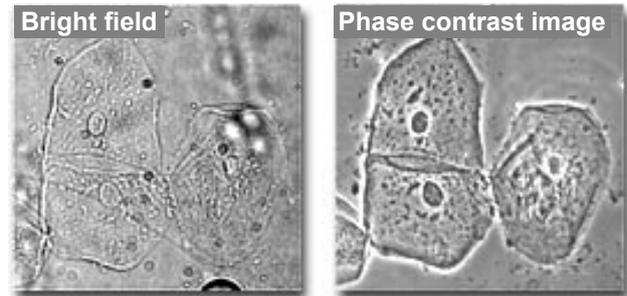
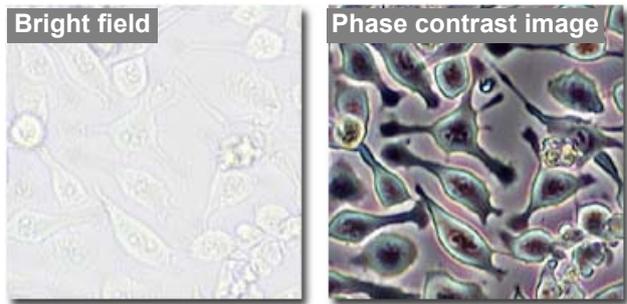
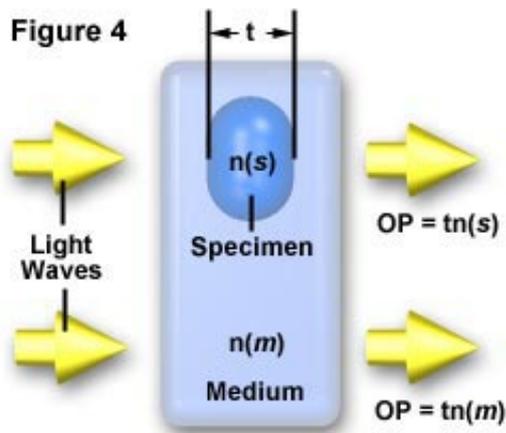


### 4.15.3 Phase Contrast Microscopy (=enhances refractive index contrast)

For transparent (e.g., organic) samples, absorption differences are usually very small. Although, the absorption contrast is weak the refractive index can still vary significantly. This creates phase changes in the transmitted optical beams due to the **optical path differences**.

In **phase contrast microscopy**, these optical path or phase differences in the transmitted beams are **converted into amplitude, i.e., intensity differences**. Thus, the contrast in transparent samples is strongly enhanced.

#### Optical Path Difference in Phase Objects



**Example:** Transparent, colorless human cells imaged with optical microscopy under differing contrast modes. Left: Bright field, Right: Phase contrast.

### Working principle (for transmission mode)

1. The conversion of phase differences into intensity differences is achieved by **separating** the directly transmitted **unscattered light** (yellow) from the **diffracted light** (red in the figure) from the sample. These two components are projected onto different locations in the objective back focal plane due to parallel beam **off-axis illumination** coming from a **condenser annulus aperture**.

2. The **directly transmitted light** appears as a ring in the back focal plane of the objective. There a ring-shaped **phase plate** of etched or semi-transparent glass is inserted, inducing a phase shift of the directly transmitted light.

The **diffracted light** is spread all over the back focal plane and is thus not much phase shifted.

Thus, the phase plate induces a **phase shift** between the direct and the diffracted light rays.

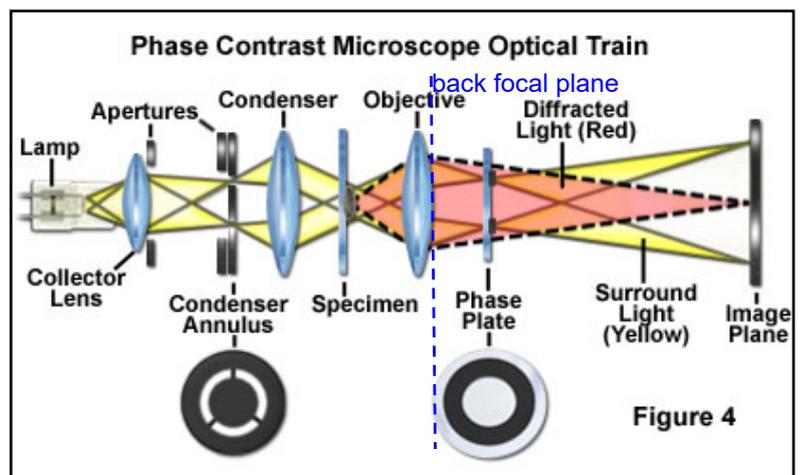
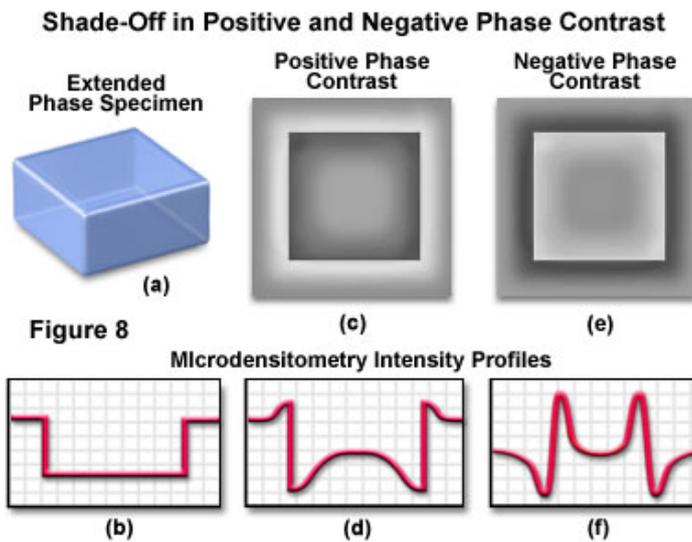
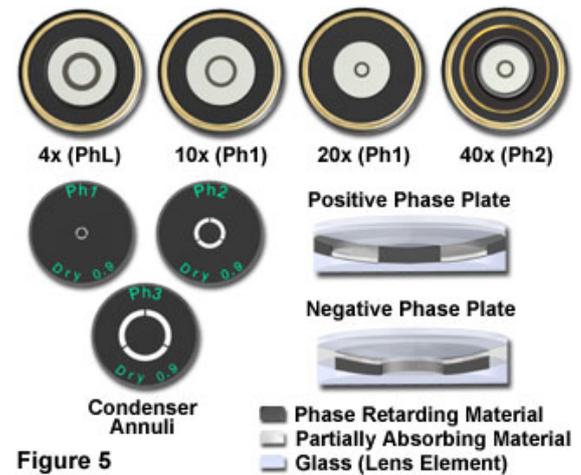


Figure 4

3. When the different light rays are combined in the image plane, the **phase difference** between these rays leads to strong **interference effects** and thus, **large intensity modulations**, which enhances the contrast from the sample. A strong contrast appears at the points in the sample where the refractive index changes.



**Objective Apertures and Phase Contrast Optics**

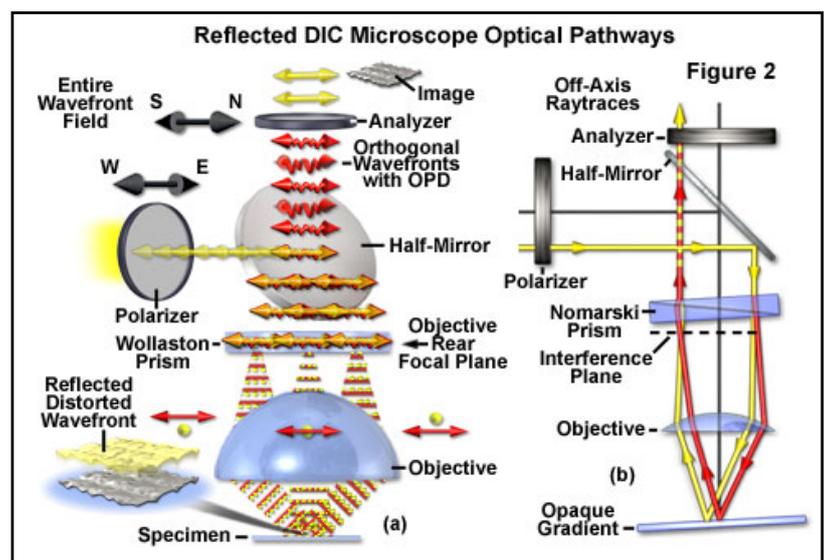


## 4.15.4 Differential Interference Contrast Microscopy (DIC/Nomarski Microscopy)

Differential interference contrast microscopy **enhances** the **sensitivity to topography** (reflection mode) or to **thickness / refractive index variations** (transmission mode) in the sample.

### Working principle:

- (1) The illumination light passes through a linear polarizer and projected through a half-mirror on the sample.
- (2) A bi-refringent **Nomarski prism** separates the light **into two orthogonally polarized components**, which are **slightly sheared** with respect to each other. The objective focuses sheared wave fronts on the surface.
- (3) The two shear waves experience varying optical **path differences as a function of surface topography** and are collected by the objective and are recombined and superimposed by the Nomarski prism.
- (4) After the Nomarski prism, the waves pass through the half-mirror and the second polarizer analyzer. Components of the orthogonal wave fronts parallel to the analyzer transmission vector to pass through in a common azimuth, and subsequently interfere in the image plane. By **adjusting the relative orientation of the polarizer and analyzer axes**, the **interference contrast** in the images **can be tuned**.



# Resulting DIC contrast

Because the **intensities** in the image are extremely sensitive to the **phase gradient** between the **orthogonally polarized beams**, the Nomarski microscope drastically **enhances the resolution in z-direction** down to a few nanometers.

Thus, the **topographic contrast** in the images is **strongly increased** and is particularly sensitive to surface steps.

**Examples:**

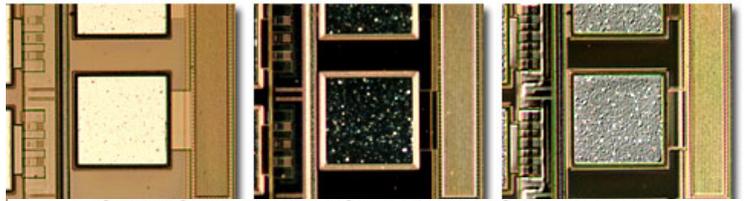
Distinction between phase contrast (PCM) and differential interference contrast (DIC):

**PCM:** Intensity  $\sim$  absolute phase differences

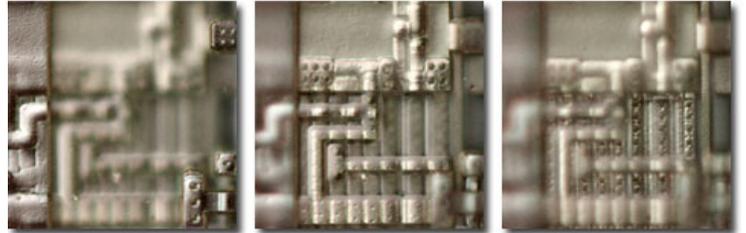
**DIC:** Intensity  $\sim$  phase gradients or  $\sim$  topography gradients

$\Rightarrow$  Thus, DIC images appear as derivative of the bright field images, as if the sample were illuminated from the side.

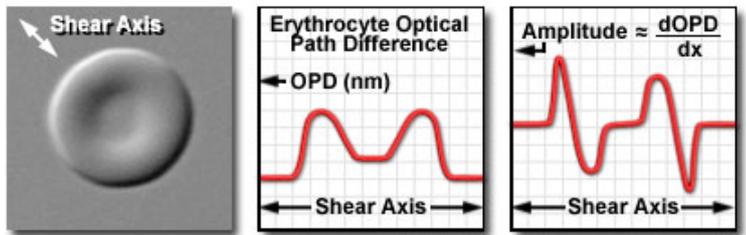
Integrated Circuit in Brightfield, Darkfield, and DIC with Reflected Light



Optical Sectioning in Reflected Light DIC Microscopy

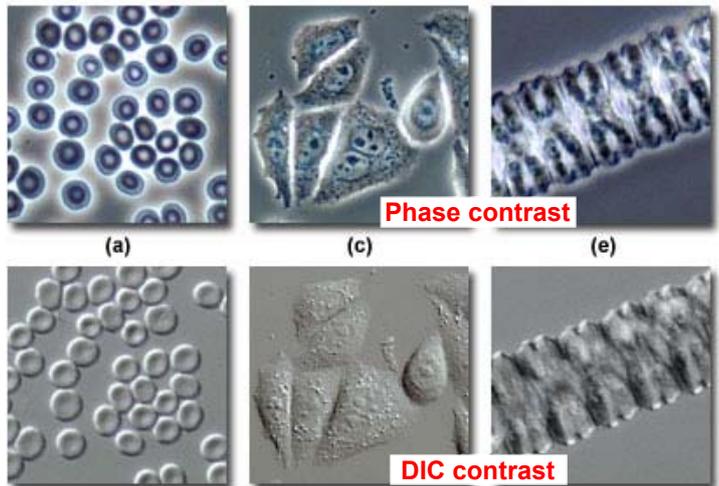
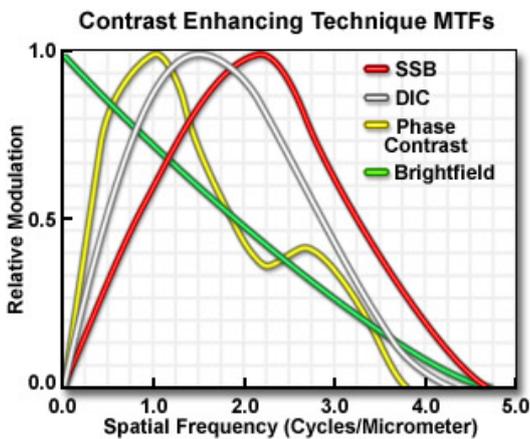


Specimen Optical Path Difference and DIC Amplitude Profile



## 4.15.5 Comparison of Different Contrast Enhancement Techniques

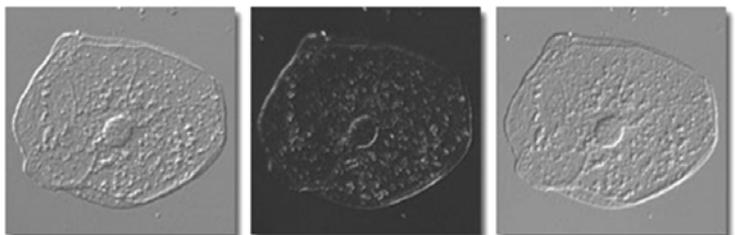
**Modulation transfer functions:**



**Effect of off-axis illumination**



**Positive and Negative Bias in Differential Interference Contrast**



## 4.16 Summary

- ❖ Two main approaches for microscopy exist: **Scanning and imaging microscopy**. These either consist of local excitation and non-local detector (scanning microscopy), or of non-local excitation and local detection (imaging microscopy).
- ❖ The **resolution** in scanning microscopy is limited by the **spot size** and **interaction volume**.
- ❖ In imaging microscopy, the **resolution is limited by two factors**:
  - (a) **Diffraction** at the objective aperture, which yields:  $r_{diff} = 0.61 \lambda / n \sin \alpha$
  - (b) **Lens aberrations**, introducing broadening
    - $r_{sph} = C_s \cdot \alpha^n$   $n \sim 3$ ,  $C_s \sim f$
    - due to spherical & chromatic aberrations  $r_{chr} = C_c \cdot dn/n \cdot \alpha^n$   $n \sim 1$ ,  $C_c \sim f$
 Other aberration effects include coma and field of view curvature.
 The **total effective resolution** is thus given by the sum  $r_{eff} = (r_{dif}^2 + r_{aber}^2)^{1/2}$
- ❖ The **depth-of-focus** decreases with increasing numerical aperture:  $DOF \approx n \cdot \lambda / NA^2$   
By recording **z-stacks** (focus through series), 3D images can be obtained.
- ❖ Due to efficient aberrations corrections, the resolution in optical microscopy is mainly diffraction limited and the highest resolution is achieved for **high numerical apertures**  $NA = n \sin \alpha$  and **short wavelength**  $\lambda$  imaging systems.  
For electron and x-ray microscopy the resolution is mainly aberration limited because of the much lower quality of the lenses.

## ❖ Point spread function and modulation transfer function

The image formation process can be described either by the

- (i) **convolution of the object with the point spread function**, or by
- (ii) convolution of the **Fourier transformation of the object** with the **modulation transfer function**, which describes the contrast modulation amplitude in the image versus that of the sample amplitude as a function of spatial frequency, i.e., periodicity.

Both functions precisely describe the performance of a microscope.

## ❖ **Contrast** is the second crucial parameter for microscopy.

It influences the "**practical**" resolution due to the visibility criterion.

- » Therefore, a **high signal-to-noise ratio** is essential high resolution imaging.  
This can be much improved by proper signal averaging and increase of integration times.
- » **Many different mechanisms contribute to the contrast** such as differences in absorption, reflectivity, topography, diffraction, composition (Z-contrast), strain, phase differences, etc.
- » For many specimen, the contrast is weak / insufficient.
- » **Contrast enhancement techniques** are therefore employed such as staining, selective etching, phase-, interference-contrast, off-axis illumination, etc. .

This modifies the modulation transfer function and enhances it at higher spatial frequencies, thus, increasing the visibility especially of small scale structures closed the resolution limit.